

1 **Harvesting Energy via Water Movement and Surface Ionics in Microfibrous Ceramic**

2 **Wools**

3 Manpreet Kaur^{a,b*}, Avinash Alagumalai^b, Omid Mahian^{c,d,e,i*}, Sameh M. Osman^f, Tadaaki
4 Nagao^{a, g*}, Zhong Lin Wang^{e,h}

5 ^aInternational Center for Materials Nanoarchitectonics (WPI- MANA), National Institute for
6 Materials Science (NIMS), Tsukuba, Ibaraki, 305-0044, Japan

7 ^bDepartment of Chemical and Petroleum Engineering, University of Calgary, 2500 University
8 Dr. NW, Calgary, Alberta, T2N 1N4, Canada

9 ^cZhejiang Provincial Engineering Research Center for the Safety of Pressure Vessel and
10 Pipeline, Faculty of Mechanical Engineering and Mechanics, Ningbo University, Ningbo
11 315211, China

12 ^dDepartment of Chemical Engineering, Imperial College London, London SW7 2AZ, UK

13 ^e Beijing Institute of Nanoenergy and Nanosystems, Chinese Academy of Sciences, Beijing,
14 100083, People's Republic of China

15 ^f Chemistry Department, College of Science, King Saud University, P.O. Box 2455, Riyadh
16 11451, Saudi Arabia

17 ^g Department of Condensed Matter Physics, Hokkaido University, Sapporo, Hokkaido, 060-
18 0810, Japan

19 ^hSchool of Materials Science and Engineering, Georgia Institute of Technology, Atlanta, GA,
20 USA

21 ⁱ Laboratory on Convective Heat and Mass Transfer, Tomsk State University, 634045, Tomsk,
22 Russia

23 Corresponding authors: Manpreet (manpreet.kaur7@ucalgary.ca); Omid

24 (o.mahian@imperial.ac.uk; omid.mahian@gmail.com); Tadaaki (nagao.tadaaki@nims.go.jp)

25 **Abstract**

26 Due to the push for carbon neutrality in various human activities, the development of
27 methods for producing electricity without relying on chemical processes or heat sources has
28 become highly significant. Also, the challenge lies in achieving microwatt-scale outputs due
29 to the inherent conductivity of the materials and diverting electric currents. To address this
30 challenge, our research has concentrated on identifying non-conductive mediums for water-

31 based using low-cost microfibrinous ceramic wools in conjunction with a NaCl aqueous
32 solution for power generation. The main source of electricity originates from the directed
33 movement of water molecules and surface ions through densely packed microfibrinous ceramic
34 wools due to the effect of dynamic electric double layer. This occurrence bears resemblance
35 to the natural water transpiration in plants, thereby presenting a fresh and straightforward
36 approach for producing eco-friendly electricity. The generator module suggested in this study,
37 measuring $12 \times 6 \text{ cm}^2$, demonstrated a noteworthy open-circuit voltage of 0.35 V, coupled
38 with a short-circuit current of 0.51 mA. Such low-cost ceramic wools are suitable for
39 ubiquitous, permanent energy sources and hold potential for use as self-powered sensors and
40 systems, eliminating the requirement for external energy sources such as sunlight or heat.

41 **Keywords:** Energy harvesting; ceramic microfibers; water evaporation; power generation;
42 self-powered systems; clean energy.

43 **1. Introduction**

44 Over the upcoming decades, easily accessible fossil energy resources are anticipated to
45 diminish significantly, exacerbated by their persistent adverse effects on the environmental ¹⁻⁴.
46 Recently renewable energy research has advanced to the forefront, and zero-emission energy
47 sources and carbon neutrality have become top priorities in energy research ⁵⁻⁹. Several
48 devices have been successfully developed utilizing battery-free, self-powered systems, such
49 as piezoelectric or triboelectric nanogenerators ¹⁰⁻¹². Indeed, the concept of self-powered
50 sensors and systems based on “Nanogenerators” was coined by Wang and Song in 2006 ^{13, 14}.
51 Most such energy-harvesting systems have limitations in regard to operational lifetimes,
52 useful geographic region, or fabrication costs ^{15, 16}. Also, their nano-fabrication processes are
53 rather demanding and fall far short of meeting industrial needs ¹⁷⁻²⁰. In recent times, there has
54 been a lot of interest in water-based power generation due to its capacity to harness clean
55 energy stored within the dynamic states of water ²¹⁻²⁶. If the energy absorbed by water could

56 be harnessed at an efficiency of 1% using water-based power technology, the potential
57 contribution could amount to approximately one-third of the world's energy consumption,
58 which is comparable to the energy output of crude oil ²⁷. However, this method typically
59 necessitates variations in elevation and is unsuitable for decentralized and widespread energy
60 extraction. In the recent past, innovative nanogenerators that capitalize on water flow have
61 emerged as a solution to overcome this limitation ²⁸⁻³¹. In such devices, electricity generation
62 is based on streaming current, in which an ionic species near the electrical double layer
63 (EDL) at the solid-liquid ³²⁻³⁵ interface carries the electricity during water movement ³⁶⁻³⁸.
64 Due to the ample availability of water resources and the increasing requirement for
65 environmentally friendly energy sources, a novel approach has been documented wherein
66 electricity generation is propelled by water. Carbon black, graphene oxide, and carbon
67 nanotubes have been studied together with their hybrid structures^{37, 39-45}. However, their
68 achieved voltages have been constrained to millivolt ranges, while their wattage outputs
69 predominantly remain within the microwatt scale. This limitation is likely due to the inherent
70 conductive properties of these materials, which tend to divert the electric currents. To address
71 this challenge, our research has concentrated on identifying non-conductive mediums for
72 water-based power generation. Our focus has shifted towards water transpiration from
73 ceramic media, as we anticipate that these materials hold the potential to deliver higher
74 voltage and power outputs in comparison to systems reliant on carbon-based components. In
75 this work, an endeavor has been made to study the generation of electrical voltage through
76 the process of water transpiration across a ceramic microfiber wool (CW), which acts as an
77 "insulator". This specific ceramic microfiber wool possesses a notable combination of
78 attributes, including a substantial surface area and a multitude of narrow spaces that facilitate
79 capillary forces. Among the numerous options available in the realm of micro-fibrous
80 ceramic materials, CW stands out due to its cost-effectiveness and wide-scale industrial

81 production. Our previous research demonstrated that the fibrous structure of CW employs
82 capillary action for efficient water evaporation, leading to swift near-room-temperature
83 distillation through solar photothermal evaporation ⁴⁶. Building on this, our current study
84 focuses on harnessing the natural movement of water through CW, driven by
85 evapotranspiration, to generate electrical power.

86 We have utilized commercially available ceramic wool (ISOWOOL Blankets 1260; Isolite
87 Insulating Products Co. Ltd) in our experiments. These ceramic wool products are widely
88 used in refractories for high-temperature equipment and are among the least expensive and
89 most mass-produced industrial materials in this category. The decision to use ceramic fibers
90 with the composition of microfibers comprising 54% SiO₂ and 46% Al₂O₃ was based on their
91 suitable properties for filtration or capillary tasks in corrosive media, elevated temperatures,
92 or pressures. Furthermore, these environmentally friendly and low cost ceramic fibers are
93 flexible, easy to handle, and can be woven into fabrics, allowing for the fabrication of
94 complex-shaped composites.

95 Through our experiments, we have demonstrated that surface-modified CW serves as a
96 proficient power generator fabricated using a device measuring 12×6 cm². Furthermore, we
97 explored the potential for voltage scaling by arranging such devices in series, leading to an
98 impressive voltage increase to 1.6 V. This level of voltage proves adequate for tasks such as
99 charging a supercapacitor to illuminate an LED or operate a motorized fan.

100 A remarkable advantage of this system is its fuel source – ionized water, which is naturally
101 evaporated, can serve as the generator's fuel. Consequently, this generator can operate
102 continuously, both during the day and night. This unique trait positions it as a valuable power
103 supply for self-sustaining systems, thereby complementing the operation of solar cells.

104

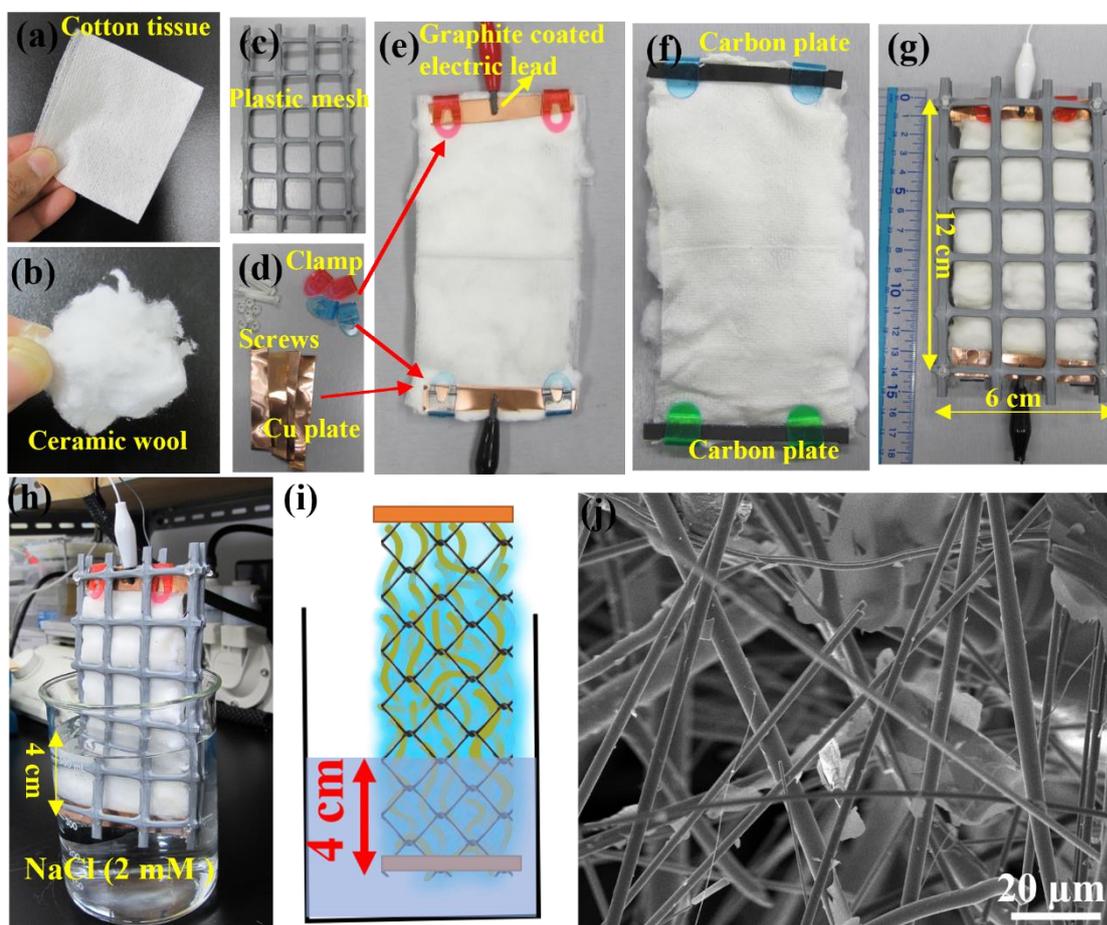
105 **2. Methods**

106 Initially, the CW underwent a cleaning process using a sodium hydroxide solution. To
107 transition the hydrophobic nature of CW to hydrophilic, hydroxyl (OH) groups were
108 introduced through the reaction with sodium hydroxide. The negative zeta potential indicates
109 the negative charges due to OH groups on the modified CW surface. In all measurements,
110 this OH modified hydrophilic CW was used, and the narrow spacing in between the fibers
111 induced a capillary force that resulted in efficient uptake and transpiration of water by this
112 material.

113 **Figure 1** provides an overview of the entire apparatus and experimental arrangement
114 employed to measure the electric potential generated through evaporation. To fabricate the
115 ceramic wool electric generator (CWEG) device, two cotton sheets measuring 12×6 cm² each
116 were prepared. Subsequently, 1,000 mg of CW was reshaped into a thin layer measuring
117 12×6 cm² with a thickness of approximately 0.2 cm. This CW layer was then placed between
118 the two cotton sheets, and this assembly was further enclosed between two plastic grids as
119 depicted in Figure 1(a-i). The entire structure was secured using plastic clamps to fasten the
120 electrodes in place. The electrodes were copper plates or carbon-glass plates (Figure 1e and
121 1h). To simplify the handling of the CWEG device, copper (Cu) plates were utilized as
122 electrodes in all conducted experiments. This device, with the Cu electrodes, was positioned
123 within a beaker to facilitate the measurement of the potential variation across the CWEG
124 sample.

125 The employed CW comprised microfibers which exhibited an average diameter of around 3.2
126 μm (as depicted in Figure 1j). By utilizing the Brunauer–Emmett–Teller method, the specific
127 surface area of the CW was determined to be 70.99 m²g⁻¹. Additionally, electrical
128 connections were established using two electrical clamps. Graphite paste was used to
129 safeguard the electrode. This paste-coated area was subsequently dried at a temperature of

130 50 °C for a duration of 3 hours. This measure was undertaken to prevent corrosion within a 2
 131 mM saline environment. **Figure S1** shows (a) Prior to and (b) Post graphite coating effect
 132 pictures of the electrode.



133
 134 **Figure 1.** Design of a ceramic wool electric generator (CWEG) (a) commercial grade cotton fiber
 135 cloth (b) ceramic wool (CW) (c) plastic grid (d) plastic clamp, plastic screw, and Cu plates (e)
 136 prepared CWEG (f) prepared CWEG using carbon glass plate (g) CWEG in a grid (h) complete
 137 CWEG setup in NaCl (i) schematic diagram of the setup for current-voltage measurement (j) SEM of
 138 CW.

139 The CWEG was introduced to a 2 mM NaCl aqueous solution. This solution covered a 4 cm
 140 segment of the CWEG, while the rest remained exposed to the surrounding atmosphere
 141 (depicted in Figure 1i). One of the electrodes positioned at the extremity of the CWEGs was
 142 submerged in the water. The water infiltrated to a level several centimeters above the water

143 surface in the beaker due to capillarity effect of filler. This capillary effect emulates the water
144 transpiration observed in plants.

145 The energy conversion performance of the CW device was determined using a VersaSTAT
146 potentiometer (VersaSTAT 4, Princeton Applied Research) under water evaporation. Before
147 the measurements, the fibers were readied with the NaCl solution for almost 1 h. All tests
148 were performed thrice to determine their repeatability. The following are the CWEG
149 dimensions; $\sim 12 \times 6 \text{ cm}^2$, thickness $\sim 0.2 \text{ cm}$, weight $\sim 1 \text{ g}$ (optimized CWEG dimensions
150 and discussed below), followed by all other experiments in this work. The voltage and current
151 values with the mean and standard deviation of five devices is shown in Table S1.

152

153 **3. Results and Discussion**

154 The working principle of the evaporative nanogenerator is shown in **Figure 2**. Wang et al.'s
155 research underscores the primary involvement of electrons in the initial formation of the
156 charge layer during the phenomenon of liquid-solid contact electrification, commonly
157 referred to as triboelectrification⁴⁷⁻⁴⁹. The nanogenerator, which relies on water evaporation,
158 utilizes this phenomenon for mechanical energy harvesting and investigating charge transfer
159 at these interfaces. These nanogenerators operate by combining contact electrification and
160 electrostatic induction, entailing surface electron/ion/material transfer and generating electric
161 potential differences upon separation in materials with contrasting triboelectric polarity^{50,51}.
162 Ceramic fibers possess several desirable qualities over the previous used nanogenerators
163 materials, rendering them appealing for filtration or capillary applications in corrosive
164 environments, high temperatures, or under elevated pressures.

165 Measured zeta potential values (discussed later) are well aligned with the origin of the
166 evaporation induced streaming potential and chemistry of NaCl on the CW surface. The zeta
167 potential values give an indication of the negative charge on the modified CW surface. The

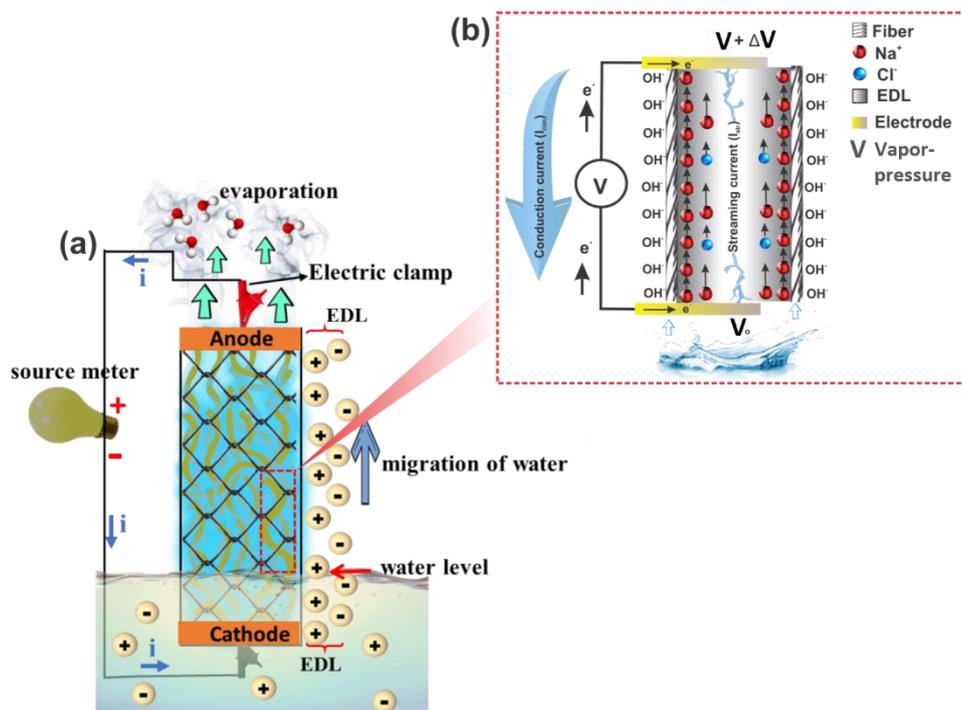
168 noticeable hydrophilic characteristics of CW, attributable to the presence of surface -OH
169 groups (as confirmed by zeta potential values), facilitate the deep penetration of water
170 molecules into the interior of the CWEG. As a result, the water adsorbed by the hydrophilic
171 nature of CW promptly wets the surface, consequently triggering surface charging. Notably,
172 during the measurement, the bottom CWEG electrode functions as the negative electrode,
173 while the top electrode serves as the positive electrode.

174 In the aqueous environment, the presence of ions initiates the development of an interfacial
175 charge layer that closely resembles an EDL when these ions interact with the hydroxyl (OH)
176 groups situated on the ceramic fibers. Counterions, such as Na^+ , exhibit a distinct preference
177 for upward migration within the mobile layer of the EDL due to the capillary-induced
178 pressure gradient ⁵². This upward migration ultimately leads to the creation of charge
179 polarization, subsequently giving rise to the generation of an induced streaming potential.
180 Chemically, this is due to the dynamic electric double layer, which is produced by the change
181 in liquid covered surface area and/or ion migration at the vicinity of the surface due to charge
182 transfer between liquid and solid ⁴⁹. The structure of the Capillary-Woven CWEG doesn't
183 inherently possess pores, but rather its porosity arises from the interstitial spaces between the
184 crisscrossed fibers. When an electrolyte solution flows through such a capillary or porous
185 medium with narrow spaces, comparable to the Debye length of the fluid, counterions tend to
186 occupy these spaces due to the overlapping EDLs.

187 In summary, upon inserting the CWEG into NaCl solution, contact electrification occurs,
188 resulting in the negatively charged CW walls and the spontaneous formation of an EDL at the
189 interface between the CW fibers and the surrounding water. This process is notably facilitated
190 by the combined influences of capillary forces and evaporation. The measured zeta potential
191 of CW is -49.09 mV. To maintain charge neutrality at the CW-NaCl interface, negatively

192 charged CW walls attract cations (Na^+), while excluding anions (Cl^-), creating the EDL. As
 193 water flows, these ions move upward with it. In evaporation-driven flow, the EDL generates
 194 a net positive charge transport, creating a streaming current and inducing a streaming
 195 potential. This potential difference between CWEG electrodes drives a continuous
 196 unidirectional electron flow, resulting in a conduction current.

197 Considering the complex fibrous CW architecture, numerous electrochemical reactions likely
 198 occur on the electrodes during the passage of current. However, for clarity, a generalized
 199 schematic illustrates electron movement from the induced cathode to the anode (Figure 2b).
 200



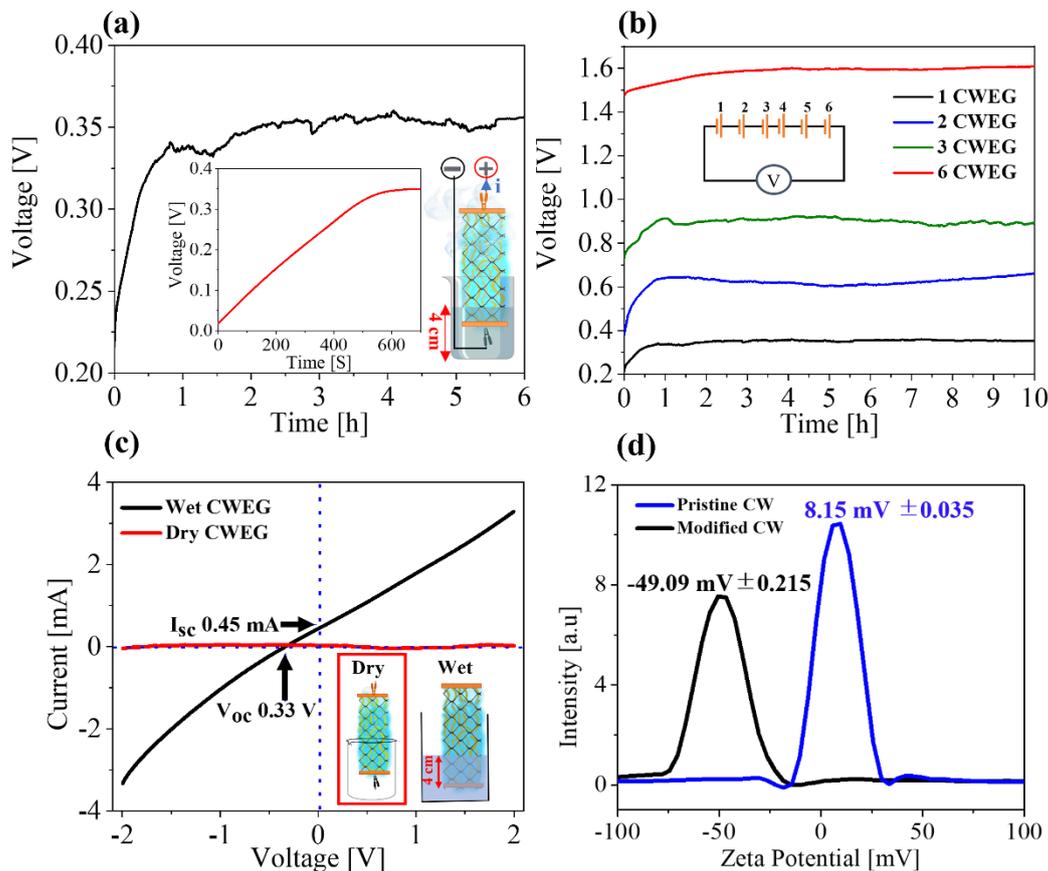
201
 202 **Figure 2.** (a) Schematic representation of the experimental setup for measuring induced potential and
 203 (b) capillary transport of ions on individual fiber, showing the specific ion distribution on the fiber.
 204 A modified hydrophilic CW with intricate fibers generates an electric potential difference
 205 between integrated electrodes as liquid flows through it. This potential difference increases

206 until reaching its peak. When dry, no open-circuit voltage (V_{oc}) is observed, but when
207 saturated with a 2 mM NaCl solution, a V_{oc} of 0.35V arises after about an hour, as water
208 reaches certain height, indicating material and solution properties influence this voltage
209 generation (**Figure 3a**). A similar experiment was performed with a carbon-glass plate
210 electrode (**Figure S2**), which gives nearly the same result as Cu-plate electrodes, indicating
211 that the redox reaction of Cu electrodes is not the main source of voltage generation. A 1- μ F
212 capacitor charges to 0.35 V in 10 min (Figure 3a, inset). Throughout all experiments, the
213 CWEG device maintains an angle of approximately 60° with respect to the water surface
214 within the beaker. This orientation permits the water to ascend to a height of roughly 4 cm
215 above the water level. Variations in the measured voltage can be attributed to the oscillations
216 in laboratory temperature and humidity during the testing period. Additionally, the suggested
217 model shows that the generated electricity can be significantly amplified by series connection
218 (Figure 3b). V_{oc} increases linearly, with slight variations due to losses stemming from wiring
219 and contact resistance. Two, three, and six samples in series can produce up to 0.61 V, 0.90
220 V, and 1.61 V, respectively.

221 Nonetheless, the induced voltage remains consistently close to 0.35 V during the entirety of a
222 6-hour test, while the short-circuit current (I_{sc}) maintains a stable value of around 0.5 mA as
223 presented in **Figure S3**. This dependable performance occurs within a laboratory
224 environment characterized by temperature variations, along with relative humidity (RH)
225 levels spanning from 45% to 55%. The initial variations in the readings can be attributed to
226 the swift advancement of the capillary front. This front eventually stabilizes as a balance is
227 achieved between water adsorption and evaporation processes. The open-circuit voltage
228 (V_{oc}) and short-circuit current (I_{sc}) are 0.35 V and 0.5mA, respectively (Figure 3c). The
229 completely dried sample before adding water into beaker shows essentially zero current.

230 Submerging the CWEG in water leads to the manifestation of an IV curve with a discernible
 231 slope and an offset.

232 Figure 3d shows that the zeta potentials of pristine and modified CW are 8.15 ± 0.035 mV,
 233 and -49.09 ± 0.215 mV, respectively, where each sample was measured 3 times and presented
 234 the mean and standard deviation values. The negative zeta potential indicates the negative
 235 charges due to OH groups on the modified CW surface⁵³. Measured zeta potential values are
 236 well aligned with the origin of the evaporation induced streaming potential and chemistry of
 237 NaCl on the CW surface as presented in **Figure 2**. The zeta potential values give an
 238 indication of the negative charge on the modified CW surface.



239

240 **Figure 3.** Electrical attributes of the CWEG device: (a) Voltage generated by a CWEG over time (in
 241 hours), while it is immersed to a depth of 4 cm in NaCl solution, (b) Voltage generated by 1 to 6
 242 CWEG devices in series mode, (c) The Performance of CWEG devices in both dry and saturated

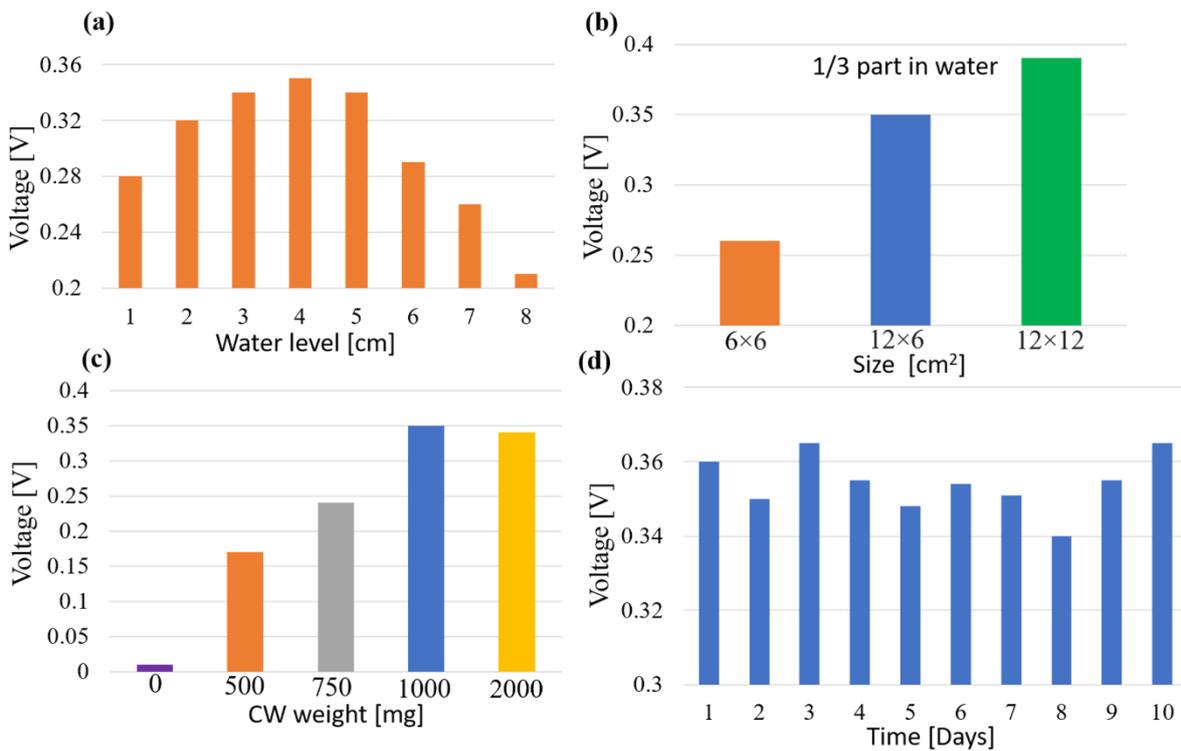
243 states within the ambient environment and (d) Zeta potential variations of pristine CW and modified
244 CW

245 Further insights into the generation of streaming potential come from additional experiments
246 using NaCl solution with different degrees of immersion of the CWEG in water (**Figure 4a**).
247 The bottom electrode of the nanogenerator was kept under the water, and the top electrode
248 was kept above it. For water depths of 1, 2, 3, 4 and 5 cm, voltages of the device were 0.27,
249 0.31, 0.33 0.35, and 0.34 V, respectively. The maximum voltage was observed, when the
250 CWEG was inserted 4 cm into the water, beyond which it decreased. With further increases
251 in the water level, the evaporation zone diminished, and the voltage decreased. Therefore,
252 placing the air-water interface near the middle of the CWEG is essential for voltage
253 generation, which effectively promotes evapotranspiration through the fibrous media.

254 On the other hand, to optimize the effect of device size on performance of the device, we
255 conducted an experiment with three different sample sizes. We first used a 6 x 6 cm² device
256 which showed 0.24 V. To increase the device performance further, we increased the length of
257 the device to 12 x 6 cm² and achieved 0.35 V. We then increased the width of the device to
258 12 x 12 cm², which further increased the voltage to 0.38 V (Figure 4b). Hence, voltage
259 generation increases with increasing CWEG area. We chose 12 x 6 cm² for the rest of our
260 measurements for ease of device setup in a laboratory beaker.

261 Next, we optimized the mass of CW sandwiched between the cotton tissues in Figure 4c and
262 Note S1. We tested voltage generation performance using cotton tissue without CW and the
263 voltage for cotton tissue only was considerably lower (0.012 V) than for the CW sandwich.
264 By sandwiching 500 mg of CW between cotton (~ 35 mg/cm³), we achieved 0.17 V, which
265 increases with higher amounts of CW, up to a maximum of 1000 mg (~ 69 mg/cm³), beyond
266 which the voltage actually decreased slightly. The higher CW fiber density (number of fibers
267 per unit length) increases the pore density, leading to higher device performance. However,

268 increasing the fiber density further increases the flow resistance, reducing the amount of
 269 evaporation. Therefore, to avoid the excess spun fibers, which diminishes the spacing
 270 efficiency of fibers, impeding ion migration, we have chosen 1000 mg CW weight (~ 69
 271 mg/cm³) for the remaining measurements.

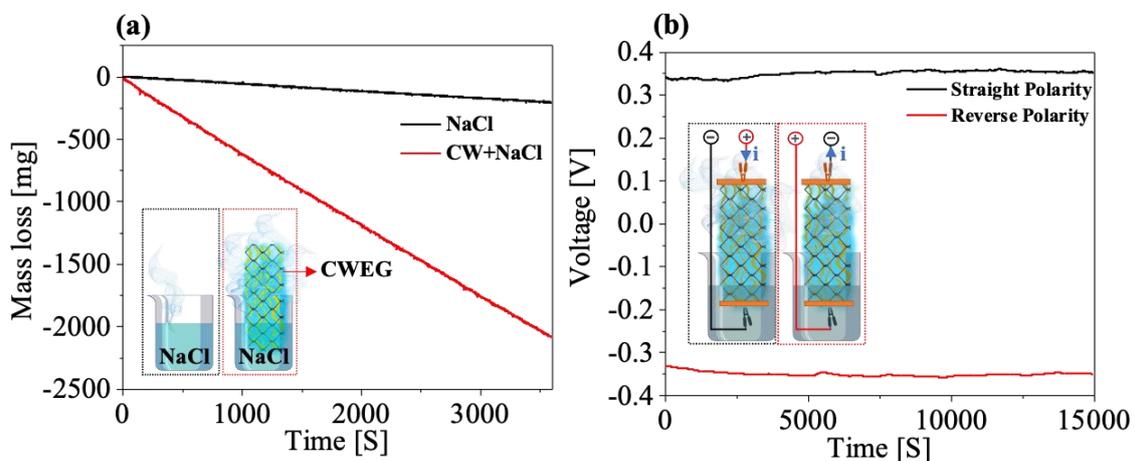


272 **Figure 4.** Voltage generation performance of CWEG with different (a) water levels (cm), (b)
 273 sample sizes, (c) CW masses sandwiched between cotton tissues, and (d) voltage generated
 274 continuously by a CWEG for 10 days, in which each cycle represents 8 hours of
 275 measurement every day.
 276

277 To assess the extended stability of the device, we conducted measurements continuously for
 278 10 days, each day spanning 8 hours (Figure 4d). Notably, the recorded voltage exhibited
 279 remarkable stability over this time frame. Throughout the course of the 10-day
 280 experimentation period, the device's performance remained nearly constant, with only
 281 marginal fluctuations in readings. These fluctuations can be attributed to alterations in
 282 ambient conditions. Consequently, the device can be regarded as a "micro 'green' power
 283 plant" capable of providing a consistent power supply with virtually negligible energy input.

284 Comparative studies on water evaporation were performed using CWEGs and NaCl solution.
 285 Evaporation of water was tested using an electric balance (AUW220D, Shimadzu). The
 286 amount of NaCl solution was fixed at 100 mL (1×10^5 mg), indicating the high evaporation
 287 efficiency of the CWEG as compared with a beaker containing only saline (**Figure 5a**). After
 288 3,600 s, the water mass loss with the CW reached 2250 mg. Therefore, due to high
 289 evaporation efficiency of the CWEG, we have observed voltages as high as 0.35 V with high
 290 stability from a 12×6 cm² device.

291 In order to further confirm the directional nature of the voltage generation, we exchanged the
 292 electrical leads, resulting in a change in polarity (Figure 5b). Upon reversing the connections,
 293 the voltage's sign also reverses, yet it retains the same magnitude. This observation leads us
 294 to the conclusion that the voltage's direction is contingent upon the direction of water flow
 295 induced by evaporation (Figure 5b).



296

297 **Figure 5.** (a) Evaporation is significantly greater with a CWEG device than without. (b)
 298 Induced potential for CWEGs connected in normal and reversed polarity (inset schematic
 299 shows sample setup).

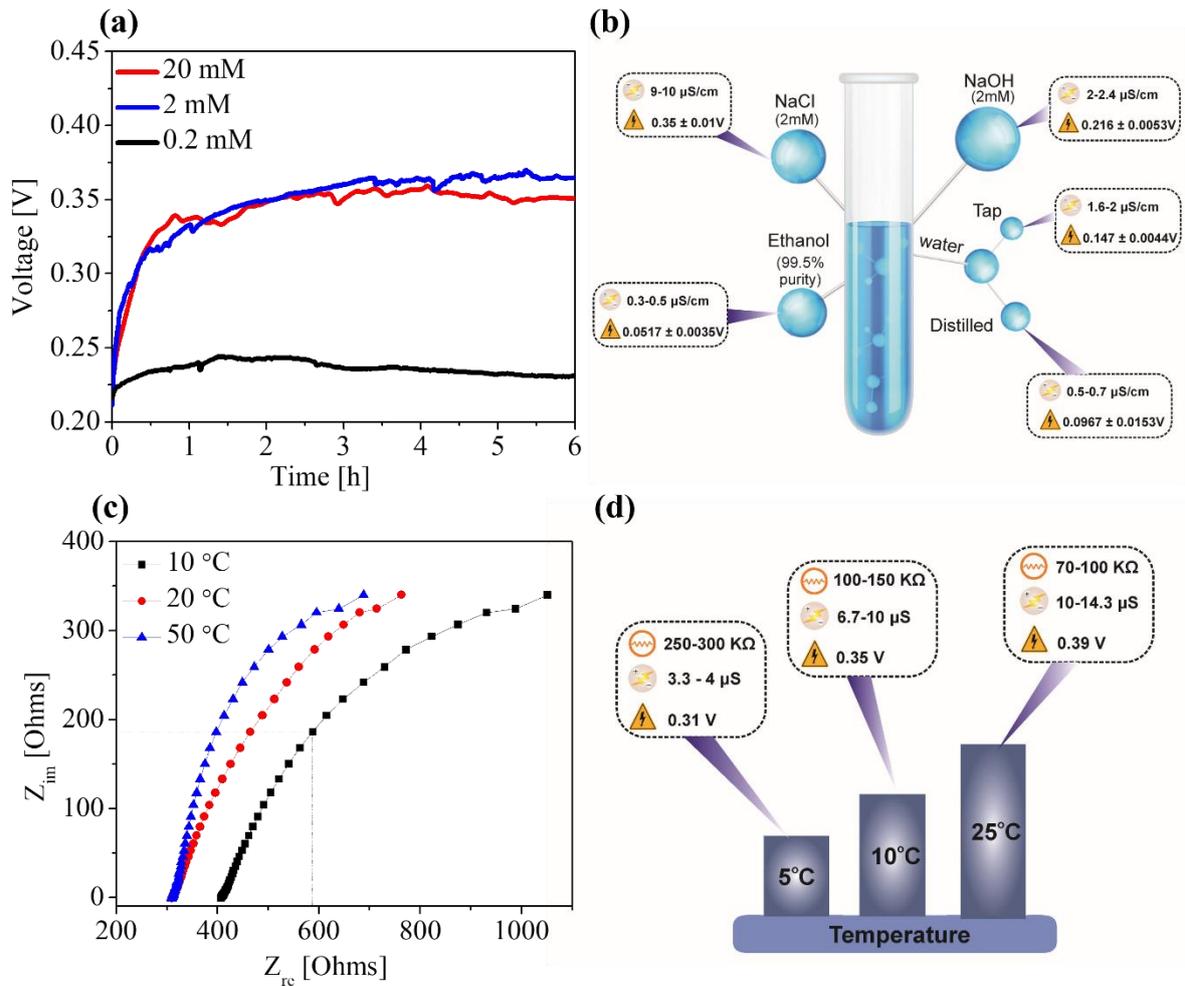
300 We conducted a comprehensive experiment to explore the relationship between induced
 301 potential and pH value (electrolyte concentration). This involved systematically adjusting the
 302 electrolyte concentration or pH value and measuring the corresponding voltage responses.

303 Three different electrolyte concentration solutions were prepared: a 0.2 mM NaCl aqueous
304 solution with a pH of ~7, a 2 mM NaCl aqueous solution with a pH of ~6 due to the increased
305 concentration of NaCl, and a 20 mM NaCl aqueous solution with a pH of ~5, reflecting a
306 further decrease in pH with increasing NaCl concentration. The induced voltage increased
307 initially as the NaCl solution molarity increased from 0.2 to 2 mM (**Figure 6a**). However,
308 when considering the 0.2 mM solution, the significantly low number of ions present might
309 not favor efficient surface ionization, leading to a diminished induced potential. In contrast,
310 the greater ion density in a 2 mM solution significantly enhances the voltage output, nearly
311 1.5 times greater compared to the 0.2 mM solution. Peculiarly, further elevating the ionic
312 concentration from 2 to 20 mM yields only minimal effects on the induced voltage. This
313 phenomenon can be comprehended through conventional electrokinetic theory. As the
314 concentration of the electrolyte rises, the thickness of the EDL diminishes. This outcome
315 leads to a reduction in the potential.^{54, 55}

316 The electrolyte solutions of different composition such as NaOH solution, tap water, distilled
317 water, and ethanol were examined whether intrinsic properties of the liquid can affect the
318 performance of the CWEG device. Furthermore, Principal Component Analysis (PCA) was
319 applied to electrolyte solutions based on voltage and conductivity variables to gain valuable
320 insights (**Figure S4**). This analysis was conducted to clarify how these electrolytes interrelate
321 concerning their electrical and conductivity traits, revealing distinct behaviors for NaOH,
322 different water types, and ethanol. A positive correlation is observed in the case of the NaOH
323 electrolyte solution, indicating heightened electrical activity. Notably, tap water and distilled
324 water fall within the same quadrant, suggesting their similarity in terms of voltage and
325 conductivity attributes within this specific feature space. The juxtaposition of positive and
326 negative values in the first and second components respectively underscores the intricate

327 influence of NaCl on the voltage and conductivity data. In contrast, ethanol showcases
328 negative scores, signifying its distinct impact on the data's variability.

329 The induced potential (Figure 6) for different solutions varies in the following manner: NaCl
330 > NaOH > tap water > distilled water > ethanol, with voltages of 0.35, 0.21, 0.15, 0.11, and
331 0.06 V, respectively. The tap water was taken from the Tsukuba, Japan municipal water
332 supply. NaCl samples generate much higher voltage values due to the high ion concentration.
333 On the other hand, a low-polarity ethanol solution does not exhibit a strong EDL due to its
334 relatively low concentration of hydrogen protons. As a consequence, the resulting voltage is
335 lower. It's worth highlighting that the streaming potential is notably more pronounced in a
336 NaCl solution. A clear correlation is seen in the difference of measured conductivity, which
337 affects the ability to respond to the moving EDL boundary.



338

339 **Figure 6.** (a) Voltage generation with respect to electrolyte concentration, (b) mean values of voltage
 340 generated from variety of electrolyte solutions and their conductivity data, (c) Effect of temperature
 341 on device performance shown by Nyquist plot (d) Values for resistivity, conductivity, and voltage
 342 with respect to temperature.

343 The charge carrier dynamics occurring at the interface between the CW and the electrolyte
 344 can be characterized through the application of electrochemical impedance spectroscopy
 345 (EIS). The electrochemical performance of the CWEG was studied using a NaCl electrolyte
 346 in a two-electrode EIS system⁵⁶. In Figure 6 (c), Z_{im} and Z_{re} represent the imaginary and real
 347 components, respectively, of the impedance (Z) as a function of frequency. Impedance is a
 348 complex quantity that describes the opposition to the flow of an alternating current in a
 349 circuit. Z_{im} (Z imaginary) refers to the imaginary component of the impedance, which

350 represents the reactive part of the impedance. It is associated with the capacitive or inductive
351 elements in the system and is typically plotted along the vertical axis of a Nyquist plot. Z_{re} (Z
352 real) refers to the real component of the impedance, which represents the resistive part of the
353 impedance. It is associated with the resistive elements in the system and is typically plotted
354 along the horizontal axis of a Nyquist plot.

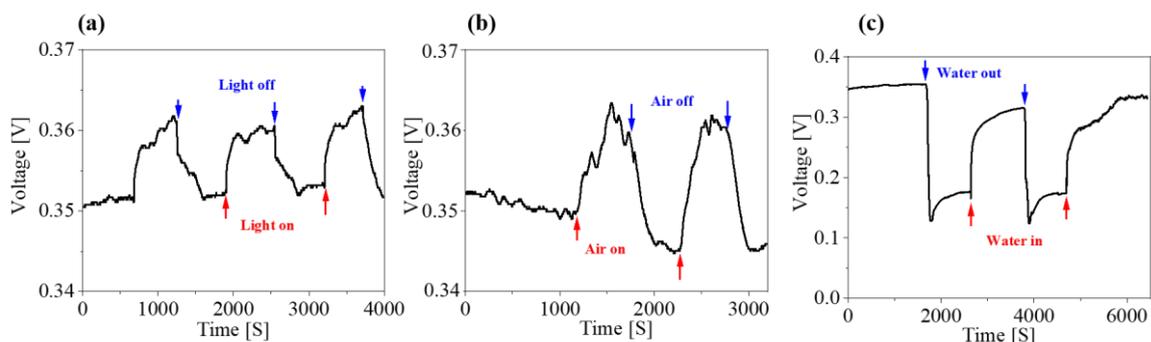
355 In the context of Figure 6(c), the Nyquist plot displays the alteration of impedance ($|Z|$) as a
356 function of frequency, with Z_{im} and Z_{re} providing insights into the reactive and resistive
357 components of the impedance, respectively. The diameter of the semicircle within the
358 Nyquist plot serves as an indicator of the charge transfer resistance from the CWEG to the
359 electrolyte, particularly in relation to the temperature of the NaCl solution.

360 The apparent trend indicates that higher temperatures substantially enhance charge transfer
361 efficiency. This is evidenced by the smaller arc observed at higher temperatures in
362 comparison to the arc displayed at lower temperatures (Figure 6c). This finding underscores
363 that the acceleration of charge transfer is markedly achieved through elevated temperatures.
364 As the temperature decreases, voltage also decreases due to a decrease in conductivity of the
365 NaCl solution (Figure 6d). The maximum V_{OC} is recorded as ~ 0.39 V at 50 °C, where the
366 effects of temperature and water evaporation raise the generated voltage significantly. The
367 performance of the device experiences a gradual decline from its optimal state as the
368 electrolyte temperature diminishes. This deterioration continues until it reaches a minimum
369 value of ~ 0.31 V at a temperature of around 10 °C. Hence, the water temperature around the
370 CWEG influences the capacity of power generation. These results suggest that the CWEG
371 would be most effective at producing electric energy in warm, dry environments.

372 Additionally, to realize the influence of temperature and water evaporation on the device's
373 performance, we conducted an additional comparison under concentrated 100-mW cm^{-2}

374 sunlight, utilizing a solar simulator (**Figure 7a**). The voltage increased and decreased as the
375 solar simulator was turned on and off, respectively. During illumination, CW absorbs light
376 and due to photothermal heat generation, evaporation is enhanced, leading to an increase in
377 voltage. The voltage returns to a low value after the light is turned off; thus the combined
378 effect of solar heating and evaporation is clearly beneficial.

379 Subsequently, in order to ascertain that evaporation indeed serves as the primary source of
380 the induced potential, the top surface of the CW was subjected to moving air at ~ 2 m/s (with
381 a temperature of around 22 °C). Air movement enhanced the evaporation rate from the CW,
382 leading to a slight increase in voltage up to ~ 0.36 V (Figure 7b). Upon ceasing the airflow,
383 the induced potential decreased to ~ 0.35 V. This clear correlation between the output voltage
384 and the rate of water evaporation underscores the significance of this process. Subsequently,
385 the behavior of the streaming voltage performance was further explored through the
386 controlled removal and injection of water (Figure 7c). The beaker was initially filled with
387 water so that the CWEG was immersed 4 cm. Then, after a few seconds, when all water was
388 removed from the beaker, voltage quickly dropped to ~ 0.13 V. The voltage difference started
389 to increase again and reached the maximum and stabilized with further water addition. This
390 behavior was highly reproducible.



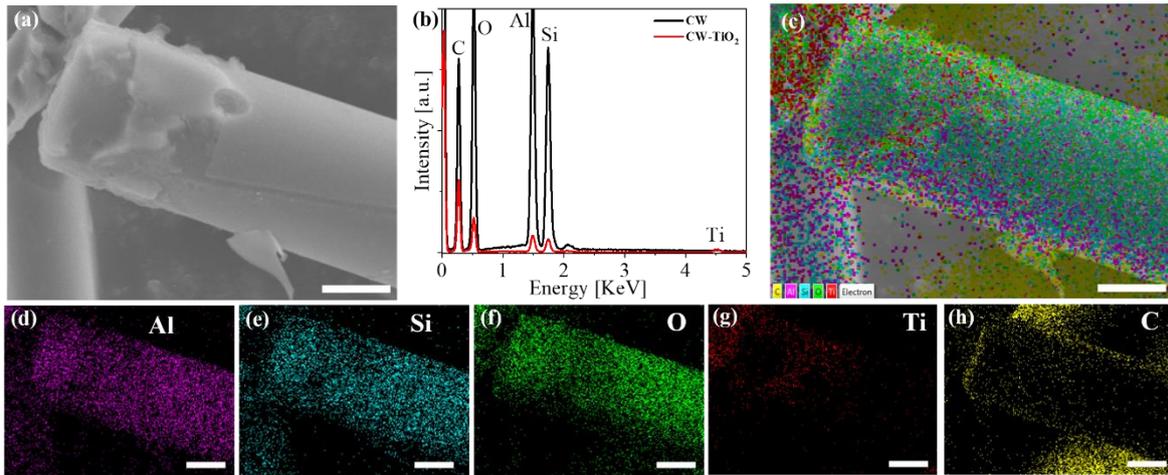
391

392 **Figures 7.** Device performance in different ambient conditions. Effect of induced potential
393 on periodic application of (a) simulated solar light (b) air flow (c) ejecting and injecting water.

394 In this work, we also studied the effect of surface modification with an ultrathin TiO₂ coating
395 on the CW fibers (CW-TiO₂). A uniform 20-nm TiO₂ coating was deposited on CW fiber
396 surfaces by atomic layer deposition (ALD) for a comparative study. The TiO₂ coating was
397 annealed at 300 °C for 3 h after deposition. Microscopic morphology of the CW-TiO₂ was
398 inspected with scanning electron microscopy (Hitachi FE-SEM SU8230) and EDX (**Figure**
399 **8**). SEM data show that CW-TiO₂ comprises fibers with a diameter of about 5 μm (Figure 8a).
400 The EDX spectrum (Figure 8b), indicates peaks of O, Al, Si, and Ti and EDX mapping
401 confirms the presence of TiO₂ on CW fibers. A carbon peak appears due to the supporting
402 carbon tape holding the sample. The overall morphology of CW-TiO₂ reveals successful
403 coating of TiO₂ on the CW. Crystallinity of the samples was characterized with X-ray
404 diffraction (XRD, RINT 2000, Rigaku). All peaks are indexed to the anatase phase of TiO₂
405 (JCPDS card No. 21-1272). The diffraction pattern of the CW exhibited a broad feature at 2
406 theta - 23°, corresponding to amorphous SiO₂ and Al₂O₃ structures⁵⁷ (**Figure S5a**). However,
407 after the CW surface modification with TiO₂, the XRD pattern of the CW-TiO₂ exhibited
408 sharp peaks corresponding to TiO₂ NPs. From the Nyquist plot (Figure S5b), it is clear that
409 TiO₂ coating exhibits a notable increase of charge transfer efficiency, resulting in a smaller
410 arc than that of bare CW, which means that charge transfer is drastically enhanced by TiO₂
411 NPs. The maximum short-circuit current density (J_{\max}) for CW and CW-TiO₂ was found to be
412 0.51 mA and 0.82 mA, respectively, while the corresponding open-circuit voltages (V_{\max})
413 were 0.35 V and 0.30 V, respectively (Figure S5c-S5d). Utilizing the formula $P_{\max} = J_{\max} \times$
414 V_{\max} , where P_{\max} represents the maximum output power density, we calculated the power
415 densities for CW and CW-TiO₂ to be 0.18 mW and 0.25 mW, respectively, based on a sample
416 size of 12×6 cm². Although the voltage decreased with the TiO₂ coating, the higher current
417 density resulted in a significant increase in the overall power density. This observation
418 indicates that while the TiO₂ coating may lead to a reduction in voltage, it concurrently

419 enhances the current output, thereby improving the overall power generation performance of
420 the CWEG.

421 Hence, the substantial increase in power output with the TiO₂ coating demonstrates the
422 potential for strategically improving CWEG performance through surface modifications with
423 appropriate materials.



424
425 **Figure 8.** EDX mapping of CW-TiO₂, (a) SEM image, (b) EDX spectrum of CW and CW-
426 TiO₂, (c), (d), (e), (f), (g) and (h) elemental mapping of Al, Si, O, Ti and C (scale bar
427 represents 2 μm).

428 The evaporative nanogenerator is capable of generating direct current (DC) electricity, which
429 can be harnessed and stored in flexible supercapacitors. These supercapacitors can
430 subsequently be interconnected to meet the energy demands of diverse electronic devices,
431 effectively providing power for various applications. **Figure 9** depicts the capillarity-coupled,
432 evaporation-induced voltage generated by a series of 6 CWEGs, yielding a maximum
433 potential of around 1.6 V. The configuration of these 6 CWEGs, along with the measurement
434 setup, is detailed in Figure 9. The power derived from these devices can be utilized to
435 illuminate a red LED, as demonstrated in **Figure S6**. Notably, the generated electricity is
436 scalable by employing devices in series, enabling it to charge a 1-μF capacitor to 1.5 V

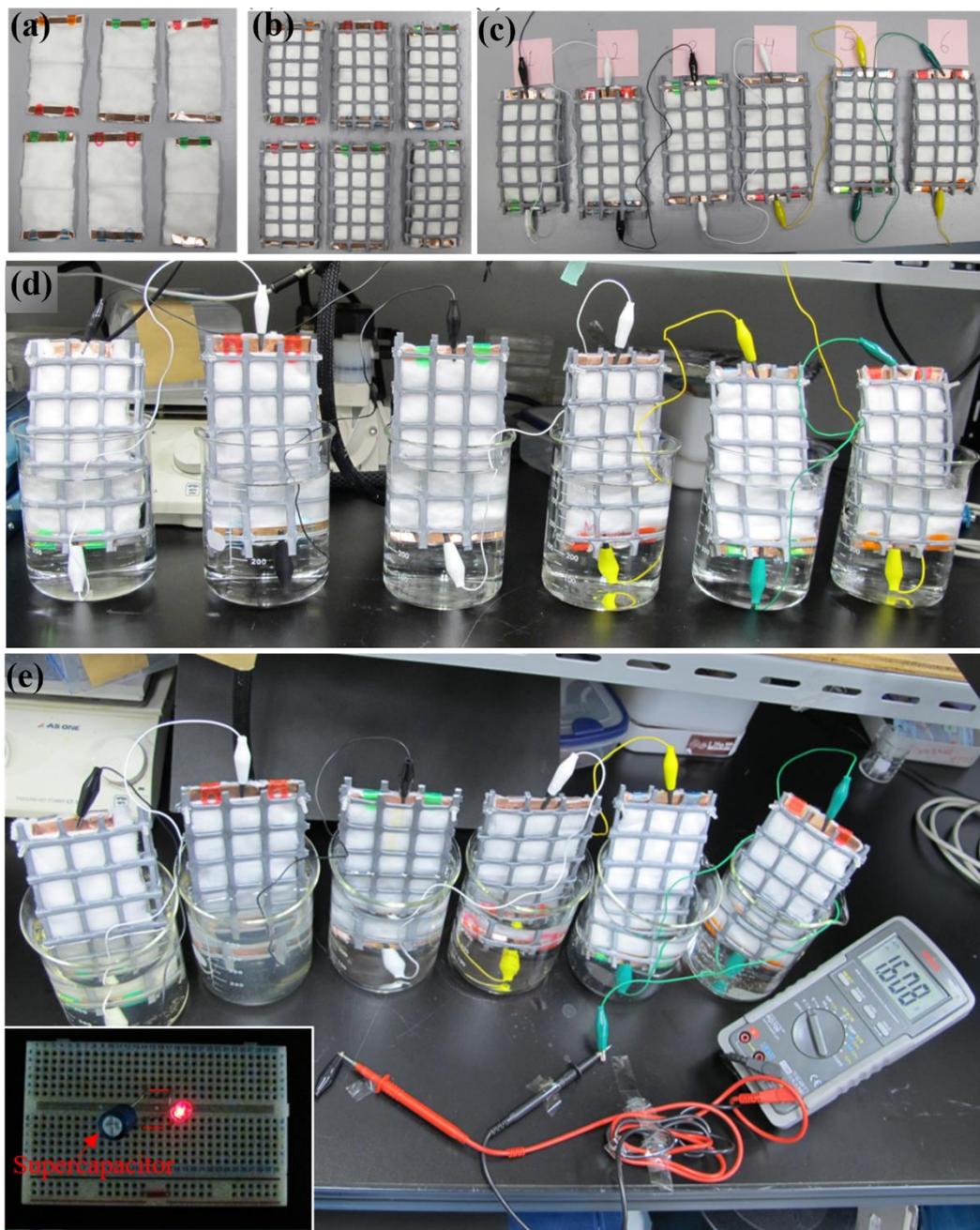
437 within a 4-hour timeframe using 6 devices in series (**Figure S7**). The energy stored through
438 this process proves sufficient to maintain the illumination of a red LED at a substantial
439 intensity for over 1 hour and to power a small DC motor for a duration of 50 seconds (Video
440 S1). The comparison of CWEG's electrical performance with latest water evaporation-based
441 power devices is presented in Table S2.

442 The outcomes of these experiments indicate that the proposed CWEG has the capacity to
443 generate sufficient power to operate small-scale electronic devices like LEDs, capacitors, and
444 micro motors. A single CWEG, measuring $12 \times 6 \text{ cm}^2$, can produce approximately 0.35 V.

445 Furthermore, the overall output performance of the system can be maximized through proper
446 selection and matching of load impedance as the real world application involves variable or
447 unpredictable load applied to the energy harvesters^{58,59}.

448 As the CWEG with size of $12 \times 6 \text{ cm}^2$ can generate $\sim 0.35 \text{ mV}$ which is slightly more than
449 the standard redox potential of copper⁶⁰. Thus, the redox reactions did not occur on the
450 copper electrodes actively. However, the redox reaction can adversely influence long-term
451 practical application of the series-connected CWEG system⁶¹. It's worth considering that
452 such redox reactions could potentially hinder the long-term practical application of series-
453 connected CWEG devices.

454 The implications of these findings are significant, suggesting the potential application of micro-
455 structured insulating materials for electricity generation. Moreover, the comparison with reported
456 phenomena, such as a single droplet lighting up LEDs or powering a light bulb⁶², highlights the
457 remarkable efficiency of water evaporation-based electricity generation.



458

459 **Figure 9.** Six CWEG devices (a) as prepared (b) in scaffolds (c) connected in series (d)
 460 connected in electrolyte solutions. (e) Scaling of CWEG performance and its use to power an
 461 LED by charging of a commercial 1- μ F supercapacitor. The inset shows a glowing LED
 462 driven through a charged supercapacitor.

463

464

465 **4. Conclusion**

466 This research demonstrates a successful utilization of water's evaporative energy to create
467 straightforward, economical devices for producing clean and sustainable power. The
468 investigation demonstrates that the evaporation of ionic water from ceramic fibers generates
469 electric voltages, and remarkably, the V_{oc} remains stable as long as the device is partially
470 immersed in water. Notably, the voltage resulting from evaporation effects can be extended
471 to around 1.6 V by connecting $12 \times 6 \text{ cm}^2$ CWEGs in series

472 These CWEGs offer an environmentally friendly solution by utilizing water for power
473 generation. The unidirectional water movement, akin to transpiration, propels efficient charge
474 transfer. This dynamic yields a voltage drop between the two electrodes that are
475 asymmetrically positioned in water and air environments. This difference in environments
476 can also result in additional chemical potential difference.

477 Our results imply that a bundle of proposed CWEG can generate sufficient power for
478 operating commercial electronic devices. The results reported here demonstrates the great
479 potential for adopting micro-structured insulating material for generating electricity for night-
480 time self-powered LEDs, capacitors, micro motors and household minigadgets,
481 environmental sensors, and so forth. This approach can improve the contact electrification
482 effect and its energy conversion efficiency in practical applications. Such low-cost ceramic
483 wools are suitable for self-powered sensors and systems that do not rely on input sources,
484 such as sunlight and heat.

485 In forthcoming research, our primary focus will be on addressing the redox reaction issue.
486 We plan to develop redox reaction-free electrodes to ensure the practical viability of the
487 CWEG system over extended operational periods. One future direction is to consider the use

488 of polymer meshes coated with conductive carbon materials to create electrodes that can
489 mitigate or eliminate these redox reactions.

490

491 **ASSOCIATED CONTENT**

492 **Supporting Information.** Figures S1-S7, Table S1-S2, Note S1 and Video S1.

493 **Funding Sources**

494 This work was funded by JSPS KAKENHI (16H06364) and CREST “Phase Interface
495 Science for Highly Efficient Energy Utilization” (JPMJCR13C3) from the Japan Science and
496 Technology Agency.

497 **Acknowledgements:** We would like to thank Dr. Takemura Taro, Miss Li Xianglan from the
498 Molecules and Materials Synthesis Platform, NIMS for Zeta Potencial measurment support.
499 The authors extend their appreciation to the Deputyship for Research and Innovation,
500 "Ministry of Education" in Saudi Arabia for funding this research (IFKSUOR3-615-1). O.M.
501 would like to thank the support of Tomsk State University Development Programme
502 (priority-2030) for this work.

503 **References**

- 504 1. H. J. Hovel, *NASA STI/Recon Technical Report A*, 1975, **76**.
- 505 2. C. J. Brabec, N. S. Sariciftci and J. C. Hummelen, *Adv. Funct. Mater.*, 2001, **11**, 15-26.
- 506 3. I. Katsouras, K. Asadi, M. Li, T. B. Van Driel, K. S. Kjaer, D. Zhao, T. Lenz, Y. Gu, P. W. Blom and
507 D. Damjanovic, *Nature materials*, 2016, **15**, 78.
- 508 4. Y. Zi, J. Wang, S. Wang, S. Li, Z. Wen, H. Guo and Z. L. Wang, *Nat Commun.*, 2016, **7**, 10987.
- 509 5. C. F. Maduko and U. B. Akuru.
- 510 6. R. Newell, D. Raimi and G. Aldana, *Resources for the Future*, 2019, 8-19.
- 511 7. A. M. Omer, *Renewable and sustainable energy reviews*, 2008, **12**, 1789-1821.
- 512 8. C. Okkerse and H. Van Bekkum, *Green Chemistry*, 1999, **1**, 107-114.
- 513 9. O. Ellabban, H. Abu-Rub and F. Blaabjerg, *Renewable and Sustainable Energy Reviews*, 2014,
514 **39**, 748-764.
- 515 10. V. Slabov, S. Kopyl, M. P. Soares dos Santos and A. L. Kholkin, *Nano-Micro Letters*, 2020, **12**,
516 1-18.
- 517 11. K. Zhao, Y. Wang, L. Han, Y. Wang, X. Luo, Z. Zhang and Y. Yang, *Nano-Micro Letters*, 2019, **11**,
518 1-19.

- 519 12. J. Yi, K. Dong, S. Shen, Y. Jiang, X. Peng, C. Ye and Z. L. Wang, *Nano-micro letters*, 2021, **13**, 1-
520 13.
- 521 13. Z. L. Wang and J. Song, *Science*, 2006, **312**, 242-246.
- 522 14. F.-R. Fan, Z.-Q. Tian and Z. L. Wang, *Nano energy*, 2012, **1**, 328-334.
- 523 15. A. J. Williams, M. F. Torquato, I. M. Cameron, A. A. Fahmy and J. Sienz, *IEEE Access*, 2021, **9**,
524 77493-77510.
- 525 16. J. Singh, R. Kaur and D. Singh, *International Journal of Energy Research*, 2021, **45**, 118-140.
- 526 17. R. Yang, Y. Qin, L. Dai and Z. L. Wang, *Nature nanotechnology*, 2009, **4**, 34.
- 527 18. M. Grätzel, *nature*, 2001, **414**, 338-344.
- 528 19. X. Wang, J. Song, J. Liu and Z. L. Wang, *Science*, 2007, **316**, 102-105.
- 529 20. Y. Jia, Y. Pan, C. Wang, C. Liu, C. Shen, C. Pan, Z. Guo and X. Liu, *Nano-Micro Letters*, 2021, **13**,
530 1-12.
- 531 21. G. J. Snyder and E. S. Toberer, in *materials for sustainable energy: a collection of peer-*
532 *reviewed research and review articles from Nature Publishing Group*, World Scientific, 2011,
533 pp. 101-110.
- 534 22. Z.-H. Lin, Y. Yang, J. M. Wu, Y. Liu, F. Zhang and Z. L. Wang, *The journal of physical chemistry*
535 *letters*, 2012, **3**, 3599-3604.
- 536 23. Y. Yang, G. Zhu, H. Zhang, J. Chen, X. Zhong, Z.-H. Lin, Y. Su, P. Bai, X. Wen and Z. L. Wang,
537 *ACS nano*, 2013, **7**, 9461-9468.
- 538 24. Z. Liu, C. Liu, Z. Chen, H. Huang, Y. Liu, L. Xue, J. Sun, X. Wang, P. Xiong and J. Zhu, 2023.
- 539 25. L. Li, S. Feng, Y. Bai, X. Yang, M. Liu, M. Hao, S. Wang, Y. Wu, F. Sun and Z. Liu, *Nature*
540 *Communications*, 2022, **13**, 1043.
- 541 26. R. Zhang, S. Wang, M. H. Yeh, C. Pan, L. Lin, R. Yu, Y. Zhang, L. Zheng, Z. Jiao and Z. L. Wang,
542 *Advanced Materials*, 2015, **27**, 6482-6487.
- 543 27. J. Yin, J. Zhou, S. Fang and W. Guo, *Joule*, 2020, **4**, 1852-1855.
- 544 28. J. Feng, M. Graf, K. Liu, D. Ovchinnikov, D. Dumcenco, M. Heiranian, V. Nandigana, N. R.
545 Aluru, A. Kis and A. Radenovic, *Nature*, 2016, **536**, 197.
- 546 29. J. Osterle, *Applied Scientific Research, Section A*, 1964, **12**, 425-434.
- 547 30. D. Burgreen and F. Nakache, *J. Appl. Mech.*, 1965, **32**, 675-679.
- 548 31. J. Yang, F. Lu, L. W. Kostiuik and D. Y. Kwok, *Journal of Micromechanics and Microengineering*,
549 2003, **13**, 963.
- 550 32. P. Král and M. Shapiro, *Physical review letters*, 2001, **86**, 131.
- 551 33. S. Ghosh, A. Sood and N. Kumar, *Science*, 2003, **299**, 1042-1044.
- 552 34. J. Liu, L. Dai and J. W. Baur, *Journal of applied physics*, 2007, **101**, 064312.
- 553 35. P. Dhiman, F. Yavari, X. Mi, H. Gullapalli, Y. Shi, P. M. Ajayan and N. Koratkar, *Nano letters*,
554 2011, **11**, 3123-3127.
- 555 36. W. Guo, C. Cheng, Y. Wu, Y. Jiang, J. Gao, D. Li and L. Jiang, *Adv. Mater.*, 2013, **25**, 6064-
556 6068.
- 557 37. G. Xue, Y. Xu, T. Ding, J. Li, J. Yin, W. Fei, Y. Cao, J. Yu, L. Yuan and L. Gong, *Nature*
558 *nanotechnology*, 2017, **12**, 317-321.
- 559 38. T. Ding, K. Liu, J. Li, G. Xue, Q. Chen, L. Huang, B. Hu and J. Zhou, *Advanced Functional*
560 *Materials*, 2017, **27**, 1700551.
- 561 39. F. Zhao, H. Cheng, Z. Zhang, L. Jiang and L. Qu, *Advanced Materials*, 2015, **27**, 4351-4357.
- 562 40. F. Zhao, Y. Liang, H. Cheng, L. Jiang and L. Qu, *Energy & Environmental Science*, 2016, **9**, 912-
563 916.
- 564 41. Y. Lu, H. Wu, Q. Yang, J. Ping, J. Wu and J. Liu, *Adv. Sustainable Sys.*, 2019, 1900012.
- 565 42. X. Gao, T. Xu, C. Shao, Y. Han, B. Lu, Z. Zhang and L. Qu, *J. Mater. Chem. A*, 2019, **7**, 20574-
566 20578.
- 567 43. J. Yin, X. Li, J. Yu, Z. Zhang, J. Zhou and W. Guo, *Nature nanotechnology*, 2014, **9**, 378-383.
- 568 44. J. Yin, Z. Zhang, X. Li, J. Yu, J. Zhou, Y. Chen and W. Guo, *Nature communications*, 2014, **5**, 1-6.

- 569 45. B. Ji, N. Chen, C. Shao, Q. Liu, J. Gao, T. Xu, H. Cheng and L. Qu, *Journal of Materials*
570 *Chemistry A*, 2019, **7**, 6766-6772.
- 571 46. M. Kaur, S. Ishii, S. L. Shinde and T. Nagao, *ACS Sustainable Chemistry & Engineering*, 2017, **5**,
572 8523-8528.
- 573 47. Z. H. Lin, G. Cheng, S. Lee, K. C. Pradel and Z. L. Wang, *Advanced Materials*, 2014, **26**, 4690-
574 4696.
- 575 48. Z. H. Lin, G. Cheng, L. Lin, S. Lee and Z. L. Wang, *Angewandte Chemie International Edition*,
576 2013, **52**, 12545-12549.
- 577 49. S. Lin, X. Chen and Z. L. Wang, *Chemical Reviews*, 2021, **122**, 5209-5232.
- 578 50. M. Kaur, S. Ishii, R. Nozaki and T. Nagao, *Scientific reports*, 2021, **11**, 1-10.
- 579 51. M. Kaur and T. Nagao, *Energy & Fuels*, 2022, **36**, 11443-11456.
- 580 52. Z. L. Wang, *Journal*, 2012.
- 581 53. M. R. Das, J. M. Borah, W. Kunz, B. W. Ninham and S. Mahiuddin, *Journal of Colloid and*
582 *Interface Science*, 2010, **344**, 482-491.
- 583 54. F. H. van der Heyden, D. Stein and C. Dekker, *Physical review letters*, 2005, **95**, 116104.
- 584 55. F. H. van der Heyden, D. J. Bonthuis, D. Stein, C. Meyer and C. Dekker, *Nano letters*, 2007, **7**,
585 1022-1025.
- 586 56. H.-C. Ho, K. Chen, T. Nagao and C.-H. Hsueh, *The Journal of Physical Chemistry C*, 2019, **123**,
587 21103-21113.
- 588 57. O. Saber and H. M. Gobara, *Egyptian Journal of Petroleum*, 2014, **23**, 445-454.
- 589 58. L.-B. Huang, W. Xu, C. Zhao, Y.-L. Zhang, K.-L. Yung, D. Diao, K. H. Fung and J. Hao, *ACS*
590 *applied materials & interfaces*, 2020, **12**, 24030-24038.
- 591 59. M. Mariello, E. Scarpa, L. Algieri, F. Guido, V. M. Mastronardi, A. Quattieri and M. De Vittorio,
592 *Energies*, 2020, **13**, 1625.
- 593 60. A. Bard, *Standard potentials in aqueous solution*, Routledge, 2017.
- 594 61. Z. Lin and Z. Yang, *Droplet*, e97.
- 595 62. W. Xu, H. Zheng, Y. Liu, X. Zhou, C. Zhang, Y. Song, X. Deng, M. Leung, Z. Yang and R. X. Xu,
596 *Nature*, 2020, **578**, 392-396.
- 597