

Supplemental Materials for 'Multipolar Phase Transition in a $4f^2$ fcc Lattice Compound PrCdNi_4 '

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I. SAMPLE CHARACTERIZATION

Backscattered electron images and powder x-ray diffraction patterns for samples A and B of polycrystalline PrCdNi_4 are shown in Figs. S1 and S2, respectively. It is noted that a polycrystalline sample from batch A was selected because it contains multiple phases. This sample includes not only PrCdNi_4 but also impurity phases such as $\text{PrNi}_2\text{Cd}_{20}$ and PrNi_5 , which is consistent with the emergence of additional peaks measured by the powder x-ray diffraction shown in Fig. S2. The atomic composition of the PrCdNi_4 phase was determined by electron-probe microanalysis (EPMA) to be $\text{Pr}_{0.99(1)}\text{Cd}_{1.01(1)}\text{Ni}_{3.87(3)}$. On the other hand, another sample from batch B is nearly single-phase, but it contains small amounts of PrNi_5 and Cd impurities. As shown in Fig. S2, there are some peaks from PrNi_5 in the powder x-ray diffraction pattern, although the peaks from $\text{PrNi}_2\text{Cd}_{20}$ are less pronounced. The composition for the PrCdNi_4 phase of sample B was determined to be $\text{Pr}_{0.97(1)}\text{Cd}_{1.03(1)}\text{Ni}_{3.84(6)}$ by EPMA. The compositions of samples A and B are similar to $\text{Pr}_{1.00(1)}\text{Cd}_{1.00(1)}\text{Ni}_{3.89(4)}$ of the sample batch for measurements, and they are close to the stoichiometric ratio compared with $\text{Pr}_{0.94(1)}\text{Mg}_{1.06(1)}\text{Ni}_{3.86(2)}$ in Ref. [1], which shows no phase transition.

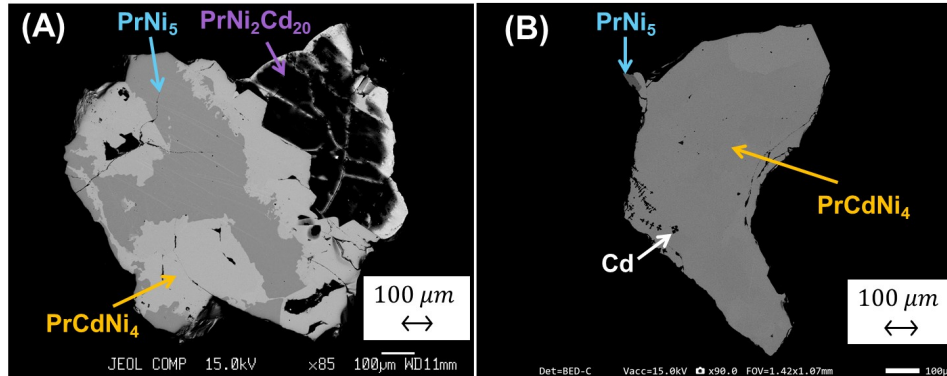


FIG. S1. Backscattered electron images of the samples A and B of PrCdNi_4 . Sample A contains PrCdNi_4 along with impurity phases, such as $\text{PrNi}_2\text{Cd}_{20}$ and PrNi_5 , whereas sample B is nearly a single phase.

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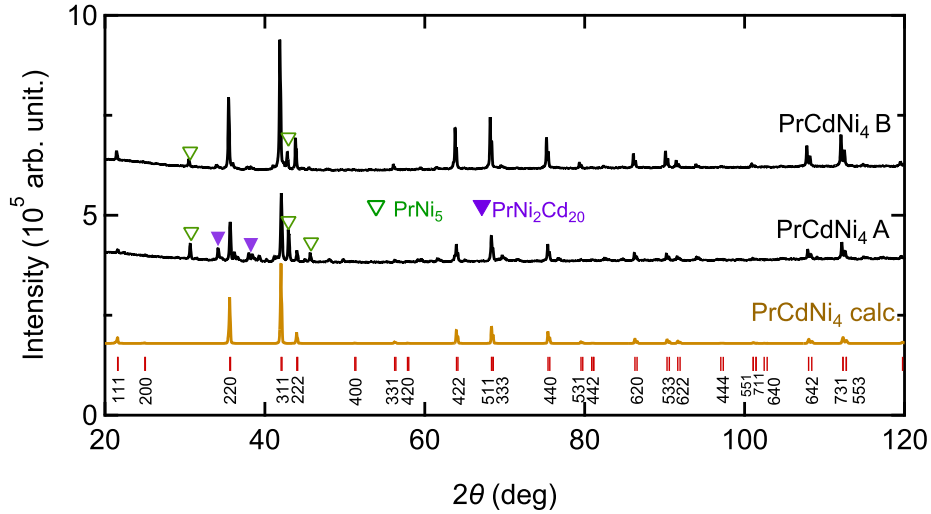


FIG. S2. Powder X-ray diffraction patterns of PrCdNi_4 (A) and (B). The lower is the simulated pattern with the cubic MgSnCu_4 -type structure. Both samples can be indexed by cubic MgSnCu_4 -type structure, while additional peaks due to impurity phases are shown with triangle.

The impurity phase of PrNi_5 was also observed in the powder neutron diffraction pattern at $T = 2.0$ K, as indicated by a star in Fig. S3. We conducted the Rietveld analysis considering the two phases, PrCdNi_4 and PrNi_5 , and determined their molar ratio to be $\text{PrCdNi}_4:\text{PrNi}_5 = 98.1(1) : 1.9(1)$. It is noted that the peaks around $2\theta = 64$ and 75 deg. were excluded from the analysis because they were due to aluminum from the sample container. Since neutrons can penetrate the samples, this result provides a reasonable stoichiometric ratio for the bulk samples in the macroscopic measurements.

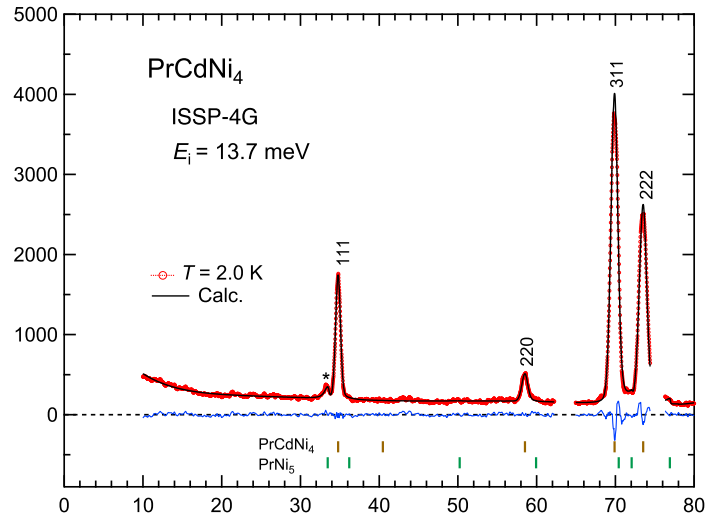


FIG. S3. Powder neutron diffraction patterns of PrCdNi_4 at $T = 2.0$ K. The black line represents the fits to the data with the Rietveld analysis. The blue line shows the difference between the data and calculation. The brown and green bars indicate the scattering angles for the nuclear Bragg peaks of PrCdNi_4 and PrNi_5 .

In Table S1, the structural parameters of PrCdNi₄ obtained from the Rietveld profile fitting analysis of neutron powder diffraction patterns for $T = 2.0$ and 0.32 K, shown with the black solid lines in Fig. 7 of the main text, are presented, along with the reliable factors for the fitting. The lattice parameter at 0.32 K decreases by 0.23% compared with the value at 273 K determined by the single-crystal X-ray analysis described in the next section.

Table S1: Structural parameters of PrCdNi₄ refined by the Rietveld profile fitting analysis of neutron powder diffraction patterns. The space group is $F\bar{4}3m$ (#216), where the Pr atoms occupy the $4a$ site at $(0, 0, 0)$, the Cd atoms the $4c$ site at $(1/4, 1/4, 1/4)$, and the Ni atoms $16e$ site at $(x_{\text{Ni}}, x_{\text{Ni}}, x_{\text{Ni}})$ with $Z = 4$.

	0.32 K	2.0 K
Lattice parameters		
a (Å)	7.1112(3)	7.1114(1)
V (Å ³)	359.61(3)	359.644(9)
x_{Ni}	0.6278(4)	0.625(2)
Reliable factor		
R_p	4.74	5.20
R_{wp}	6.24	6.75
R_e	5.40	5.36
S	1.15	1.26

II. SINGLE-CRYSTAL X-RAY STRUCTURAL ANALYSIS

To identify the crystal structure, the single-crystal x-ray structural analysis was performed at 273 K with Mo $K\alpha$ radiation, $\lambda = 0.071073$ nm, monochromated by a multilayered confocal mirror using a Bruker APEX-II ULTRA CCD area-detector diffractometer at N-BARD, Hiroshima University. Crystallographic parameters and details for the measurement, data collection, and refinement of the single-crystal X-ray diffraction experiment are described in Tables S2 and S3.

Table S2: Information of the measurement, data collection, and refinement of the single-crystal x-ray diffraction experiment for PrCdNi₄.

Crystal system	Cubic MgSnCu ₄
Space group	$F\bar{4}3m$ (#216)
a (Å)	7.12758(4)
V (Å ³)	362.098(3)
Z	4
Dimensions (mm ³)	0.21 × 0.14 × 0.08
Temperature (K)	293
Radiation, λ (nm)	Mo- $K\alpha$, 0.071073
2θ range (°)	9.908 – 89.674
μ (mm ⁻¹)	39.061
Data collection	
Measured reflections	15017
Unique reflections	186
h	$-14 \leq h \leq 14$
k	$-14 \leq k \leq 14$
l	$-14 \leq l \leq 14$
Refinement	
$\Delta\rho_{\max}/\Delta\rho_{\min}$ (eÅ ⁻³)	1.50/−1.62
GoF	1.257
Reflections/parameters	186/7
R_1	0.0157
wR_2	0.0412

Table S3: Crystallographic parameters for PrCdNi₄ determined at 273 K. U_{eq} is the isotropic displacement parameter defined as 1/3 of the trace of the orthogonalized U_{ij} tensor.

Cubic MgSnCu ₄ -type						
Space group: $F\bar{4}3m$ (#216)						
$a = 7.12758(4)$ Å, $V = 362.098(3)$ Å ³ , $Z = 4$						
Atom	Site	x	y	z	Occ.	U_{eq} (Å ²)
Pr	4a	0	0	0	1	0.0071(2)
Cd	4c	0.25	0.25	0.25	1	0.0041(2)
Ni	16e	0.62376(5)	0.62376(5)	0.62376(5)	1	0.0052(2)

III. PHONON CONTRIBUTION TO THE SPECIFIC HEAT

Figure S3 shows the temperature dependence of the specific heat $C(T)$ of PrCdNi_4 from 2 K to 300 K. The $C(T)$ data increase with elevating temperatures and approach the Dulong-Petit value of $3nR = 149.7 \text{ J/K mol}$, where $n = 6$ is the number of atoms per formula unit and R the gas constant. The inset displays the C/T^3 plots with the logarithmical scale. The C/T^3 data increase on cooling. The (red) solid curve is a fit to the C/T^3 data between 50 and 300 K by adopting the Debye model for acoustic phonon modes [2] with the Debye temperatures of $\theta_D = 270.7(6) \text{ K}$. Since the acoustic phonon contribution to $C(T)$ is proportional to T^3 at lower temperatures, the calculated C/T^3 curve approaches a constant on cooling below 20 K. Thereby, the difference between the data and the calculated curve for $T < 50 \text{ K}$ is ascribed to the electronic and magnetic contributions due to, respectively, the conduction electrons and the $4f^2$ electrons of the Pr ion. The value of $\theta_D = 270.7(6) \text{ K}$ is lower than that of $\theta_D = 300.3(6) \text{ K}$ for an isostructural LaMgNi_4 [1]. This is reasonable because the molecular weight of PrCdNi_4 is much heavier than that of LaMgNi_4 .

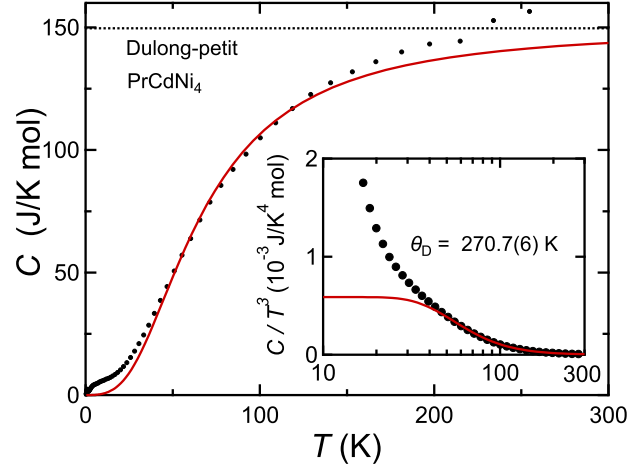


FIG. S4. Temperature variation of the specific heat of PrCdNi_4 . The inset shows the C/T^3 plot with respect to the logarithmic temperature scale. The (red) solid lines are simulated by adopting the Debye model with a Debye temperature of $\theta_D = 270.7(6) \text{ K}$.

[1] Y. Kusanose, T. Onimaru, G.B. Park, Y. Yamane, K. Umeo, T. Takabatake, N. Kawata, and T. Mizuta Jpn. J. Phys. Soc. Jpn. **88**, 083703 (2019).

[2] C. Kittel, Introduction to Solid State Physics, 8th ed., John Wiley & Sons., 2005, p.112, Chap. 5.