

1 **Transition of topological Hall effect for tetragonal Heusler**
2 **Mn₂PtSn thin film**

3
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8
9 **Abstract**

10 **The large topological Hall effect is observed in Mn₂PtSn epitaxial thin films. Non-**
11 **hysteretic topological Hall resistivity is attributed to the canted spins below the**
12 **reorientation temperature, while the hysteretic topological resistivity at zero field**
13 **vicinity captures the trend of antiskyrmion formation. A decrease in thickness**
14 **enhances the contribution of dipolar interaction, leading to the additional**
15 **antiskyrmion-type signal above the reorientation temperature. The amplitudes of**
16 **these topological signals are strongly modulated by the film thickness, providing**
17 **pathways for developing antiskyrmion hosting media via film engineering.**

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20 Topological magnetic textures possessing the Berry curvature affect the
21 electronic structure of the host material, leading to distinctive transport properties
22 favorable for future high-performance spintronic devices. [1,2] The topological Hall
23 effect (THE) is a transport property in which itinerant electrons are driven in the direction
24 of the Hall effect. [3,4] The study of THE has been intrinsically linked to magnetic
25 skyrmions, nanometric bubble domains in which intertwined wrapping spins result in a
26 unit integer topological charge. [5] Magnetic skyrmions were initially discovered in non-
27 centrosymmetric crystals with Dzyaloshinskii–Moriya interactions (DMI), [2,3,6,7] and
28 further explored in centrosymmetric systems with frustrated magnetic exchange
29 interactions. [8,9]

30 Of late, Mn-based tetragonal Heusler alloys have attracted attention as
31 alternative skyrmion-hosting materials. Antiparticles of skyrmions, known as
32 antiskyrmions (aSKs), [10,11] were observed in $\text{Mn}_{1.4}\text{Pt}_{0.9}\text{Pd}_{0.1}\text{Sn}$ in a broader field and
33 temperature range. [12] This compound belongs to the D_{2d} symmetry class, where in-
34 plane anisotropic DMI cause the stability of the robust aSK phase [13]. Studies on
35 analogous Heusler compounds reported so far include discussions on THE being linked
36 to aSK stabilizations, for example, bulk Mn_2PtSn [14] and $\text{Mn}_{1.4}\text{Pt}_{1-y}\text{Rh}_y\text{Sn}$. [15], and thin
37 films of Mn_2PtSn [16], $\text{Mn}_{2-x}\text{PtSn}$ [17,18], and Mn_2RhSn . [19,20]. Note that for such
38 anti(ferri)-magnetic materials with canted spin configurations, THE can arise in
39 noncoplanar spin structures without links to skyrmions. [21,22,23] Vistoli et al. recently
40 observed large THE in a canted antiferromagnet $\text{Ca}_{1-x}\text{Ce}_x\text{MnO}_3$, [24] and reported that
41 non-“skyrmionic” bubble domains were responsible for THE due to the canted spin
42 gradient [25]. For materials with canted spin configurations, the physical origin of THE
43 needs to be carefully explored.

44 In this letter, we report the experimental observation of a large THE in a Heusler
45 Mn_2PtSn epitaxial thin film over a wide range of temperatures, magnetic fields, and
46 thicknesses. We propose a discussion that focuses on the hysteretic properties of THE and
47 access the underlying magnetic structures.

48

49 Mn_2PtSn thin films (40 nm ~ 100 nm) were grown on MgO (001) substrates with
50 a 20 nm thick MgO buffer layer, using dc magnetron sputtering from a $\text{Mn}_{50}\text{Pt}_{25}\text{Sn}_{25}$ alloy
51 target. These films were deposited at ambient temperature and post-annealed at 500 °C
52 for 30 min. The crystal structures of the Mn_2PtSn films were characterized using X-ray
53 diffraction (XRD, Rigaku) with Cu-K α radiation ($\lambda = 1.5406 \text{ \AA}$). Atomic-scale crystal
54 structures were identified using transmission electron microscopy (TEM, FEI).
55 Magnetization measurements were performed by a vibrating sample magnetometry
56 technique using a magnetic property measurement system (MPMS, Quantum Design).
57 The transport properties were measured by a four-probe method using a physical property
58 measurement system (PPMS, Quantum Design). The variation in magnetic field between
59 MPMS and PPMS is estimated below $\pm 0.1 \text{ mT}$.

60

61 The Mn_2PtSn tetragonal crystal structure is classified in the space group $I\bar{4}m2$,
62 no. 119, as shown in Fig. 1(a). [16] The clear (110) and (220) peaks in the out-of-plane
63 XRD spectra in Fig. 1(b) support the formation of this inverse Heusler compound. The
64 cross-sectional TEM images in Fig. 1(c) capture the epitaxial growth of Mn_2PtSn crystals
65 for all thicknesses from 40 nm to 100 nm. The sharp spotty patterns of the integrated
66 nanobeam electron diffraction (NBDs), shown in Fig. 1(d), were also identified by the

67 $I\bar{4}m2$ symmetry. The lattice parameters are $a = 4.49 \pm 1\text{\AA}$ and $c = 6.08 \pm 2\text{\AA}$ for all
 68 thicknesses.

69 Figures 1(e) and (f) show the temperature dependences of longitudinal resistivity
 70 ρ_{xx} and magnetization M , respectively. The samples were field-cooled in a
 71 perpendicular field of 4 T to investigate saturated magnetization state. The kinks of ρ_{xx}
 72 and M mark the spin reorientation temperature T_s from the noncoplanar (NCP)
 73 ferrimagnet into coplanar (CP) ferrimagnets. [12] The Curie temperatures T_c are
 74 estimated as $T_c \sim 350$ K from the gradient changes of M above room temperature.

75

76 The total Hall resistivity ρ_{xy} is composed of the ordinary Hall effect (OHE)
 77 ρ_{xy}^{OHE} , the anomalous Hall effect (AHE) ρ_{xy}^{AHE} , and additional THE ρ_{xy}^{THE} , as $\rho_{xy} =$
 78 $\rho_{xy}^{\text{OHE}} + \rho_{xy}^{\text{AHE}} + \rho_{xy}^{\text{THE}}$. [2] The OHE is expressed as $\rho_{xy}^{\text{OHE}} = R_0 H$ with Hall coefficient
 79 R_0 . The AHE, dominated by intrinsic and side-jump scattering contributions [26,27], is
 80 rewritten as $\rho_{xy}^{\text{AHE}} \sim S_A \rho_{xx}^2 M$ with scaling coefficient S_A . Hence, the remaining THE
 81 contribution is simplified as follows.

$$82 \quad \rho_{xy}^{\text{THE}} = \rho_{xy} - R_0 H - S_A \rho_{xx}^2 M. \quad (1)$$

83 In Fig. 2(a), the hysteresis curves of ρ_{xy} , ρ_{xx} , $\rho_{xy}^{\text{OHE}} + \rho_{xy}^{\text{AHE}}$, and ρ_{xy}^{THE}
 84 measured under a perpendicular field are plotted for a 40 nm thick sample at 150 K. The
 85 $\rho_{xy}^{\text{OHE}} + \rho_{xy}^{\text{AHE}}$ curve deviates from the ρ_{xy} curve below $H < 2$ T, with a clear
 86 emergence of THE. The ρ_{xy}^{THE} curve shows sign inversions between the positive and
 87 negative fields as a feature of the inversed topological flux. [17,28] We obtained the
 88 maximum THE signal as $\rho_{xy,\text{max}}^{\text{THE}} = 0.57 \mu\Omega \text{ cm}$ for a 40 nm thick sample.

89 Nayak et al. reported that wide-field range THEs are primarily attributed to the
 90 noncoplanar canted spins in analogous ferrimagnetic compounds, [12,15] and

91 antiskyrmions should be linked at zero field vicinity, appearing as the hysteretic
 92 component of THE. To present the essence of these discussions, we introduce a process
 93 to separate the non-hysteretic and hysteretic topological contributions described below.

94 The non-hysteretic component of THE labeled as ρ_{xy}^{tTHE} can be extracted by
 95 adding ρ_{xy}^{THE} to change the field-sweep direction from +4 T to -4 T ($\rho_{xy,\text{dec}}^{\text{THE}}$) and
 96 change the field-sweep direction from -4 T to +4 T ($\rho_{xy,\text{inc}}^{\text{THE}}$), as follows.

$$97 \quad \rho_{xy}^{\text{tTHE}} = (\rho_{xy,\text{dec}}^{\text{THE}} + \rho_{xy,\text{inc}}^{\text{THE}})/2. \quad (2)$$

98 In addition, the hysteretic component (ρ_{xy}^{hTHE}) can be calculated by making a difference
 99 as

$$100 \quad \rho_{xy}^{\text{hTHE}} = (\rho_{xy,\text{dec}}^{\text{THE}} - \rho_{xy,\text{inc}}^{\text{THE}})/2. \quad (3)$$

101 The quantifications of ρ_{xy}^{tTHE} and ρ_{xy}^{hTHE} are shown in Figs. 2(b), (d), and (f), for 40 nm
 102 thickness at 150 K and 50 K, and for 100 nm thickness at 50 K, respectively. All ρ_{xy}^{tTHE}
 103 show a similar trend as ρ_{xy}^{THE} shown in Figs. 2(a), (c), and (e). In Fig. 2(d), finite ρ_{xy}^{hTHE}
 104 appears as a positive peak signal near the zero field, corresponding to a converse
 105 hysteresis loop in the THE curve. At a converse hysteresis, the sign inversion of ρ_{xy}^{THE}
 106 occurs after H switches to the opposite direction. The sign of ρ_{xy}^{THE} is determined by
 107 the polarity of the topological charge $Q = mp = \pm 1$, which reflects the direction of out-
 108 of-plane magnetization $p = \pm 1$ and a structure parameter called the vorticity m . [2,29]
 109 The positive ρ_{xy}^{hTHE} indicates that Q will have same sign as H with wide-range hump-
 110 shaped features in ρ_{xy}^{tTHE} . It is difficult to distinguish any topological domains from
 111 canted spins in such conditions, and we cannot insist on antiskyrmion formations only
 112 from these transport properties.

113 On the other hand, ρ_{xy}^{hTHE} shown in Fig. 2(d), obtained for a 100 nm thickness
 114 at 50 K, appears to be a negative dip $\rho_{xy}^{\text{hTHE}} < 0$ as an inverse hysteresis in the THE loop.

115 In the same manner, this trend of ρ_{xy}^{hTHE} is considered to reflect the polarity inversion of
 116 Q . Since the field direction stays constant during this process, i.e., $p = \text{const.}$, the sign
 117 inversion of vorticity m is implied. Sivakumar et al. [20] reported such inversion of
 118 vorticity m without field switching indicated antiskyrmion formations, because the
 119 vorticity in antiskyrmions considers opposite sign as $m = -1$ to that of normal
 120 skyrmions as $m = +1$, [11,29,30]. Therefore, this negative ρ_{xy}^{hTHE} is considered to be a
 121 sufficient condition of antiskyrmion formations in Mn-based Heusler thin films. It should
 122 be noted that ρ_{xy}^{hTHE} does not represent THE itself induced by antiskyrmions. The total
 123 ρ_{xy}^{THE} in Fig. 2(e) shows asymmetric behavior with respect to field inversion due to the
 124 polarity switching.

125 The other notable features are emergences of zero magnetic field topological
 126 Hall signals in Figs. 2(d) and (f). The previous report of Mn_2PtSn thin film was focused
 127 on nonzero magnetic field THE, [16] where stabilizations of antiskyrmions were observed
 128 using direct imaging techniques. [12] In these reports, the helical magnetic phases were
 129 dominant at zero field vicinities, and THE associated with antiskyrmions were discussed
 130 mainly under finite field conditions. On the other hands, zero field topological Hall
 131 signals were also reported for the recent studies of thin film Mn_2RhSn [20] and Pd-doped
 132 $\text{Mn}_{1.5}\text{PtSn}$ [28] in conditions with low temperature range where relevant dipolar
 133 interaction increases. We next present systematic results of ρ_{xy}^{THE} and ρ_{xy}^{hTHE} for
 134 different temperature, field, and sample thickness to investigate the relation between
 135 antiskyrmion formations and thin film properties.

136

137 Figures 3(a) ~ (f) show the temperature and field dependences of
 138 $(\rho_{xy}^{\text{tTHE}}, \rho_{xy}^{\text{hTHE}})$ for 100 nm, 60 nm, and 40 nm thin films. The non-hysteretic ρ_{xy}^{tTHE} of

139 all thicknesses (Figs. 3(a), (c), and (e)) reproduces the bulk nature of ferrimagnetic
140 Mn_xPtSn . [12,15] The large and constant THE signals appear below the spin reorientation
141 temperature $T < T_s$, and immediately disappear above $T > T_s$. These steep changes in
142 ρ_{xy}^{tTHE} in concert with the magnetic phase transition from the NCP to CP phases are
143 common features of the canted spin textures of ferrimagnetic Mn_xPtSn , independent of
144 any antiskyrmions.

145 On the other hand, the hysteretic ρ_{xy}^{hTHE} captures a trend of antiskyrmion
146 formations, as discussed above, and can be modulated by film thickness. For the 100-nm-
147 thick film shown in Fig. 3(b), the negative ρ_{xy}^{hTHE} , i.e., inverse hysteresis, appears below
148 $T < T_s$. The absolute amplitude of ρ_{xy}^{hTHE} decreases with temperature and reaches zero
149 around T_s . A small positive ρ_{xy}^{hTHE} remains above $T > T_s$ subsequently. The finite
150 ρ_{xy}^{hTHE} above T_s may be attributed to the pinned magnetic domains at zero field vicinity
151 due to the crystal incompleteness.

152 For the thinner 60 nm film in Fig. 3(d), on the other hand, the positive ρ_{xy}^{hTHE}
153 appears below $T < T_s$, and vanishes above $T > T_s$. This trend is similar to that of ρ_{xy}^{tTHE} ,
154 as the phase transition from the NCP phase to the CP phase without antiskyrmion
155 formation.

156 For the thinnest 40 nm film shown in Fig. 3(f), the positive ρ_{xy}^{hTHE} appears
157 below $T < T_s$ similarly to results of the 60 nm film. Additionally, antiskyrmion-type
158 $\rho_{xy}^{\text{hTHE}} < 0$ is observed above $T > T_s$, where no antiskyrmion texture has been reported
159 in bulk compounds. [15] The finite THE above T_s may be attributed to the additionally
160 stabilized antiskyrmions for thin-film engineering, also reported for MnSi and FeGe
161 compounds in cases of skyrmions. [31,32]

162 The modulation of ρ_{xy}^{hTHE} by film thickness feasibly reflects the variations in
163 the dipolar energy, which plays a critical role in antiskyrmion formation. [33] Our 100
164 nm film is thick enough to reproduce the bulk nature, leading to antiskyrmion-type signals.
165 The extension of antiskyrmion-type topological signal to zero field condition is
166 considered due to decrease in perpendicular anisotropy in thin film samples, where the
167 relevant dipolar energy will further increase from bulk samples. Once the thickness
168 decreases, the net magnetization decreases and the contribution of dipolar energy
169 becomes weaker, as is observed with the disappearance of negative ρ_{xy}^{hTHE} below 60 nm.
170 Additionally, the formation of a metastable antiskyrmion phase at the limit of a weaker
171 magnetic anisotropy was reported, [34] in which dipolar interactions are relatively
172 dominant in domain formations. Such an energy limit is observed at a thickness of 40 nm
173 above $T > T_s$, where an increase in temperature promptly decreases the magnetic
174 anisotropy, and the contribution of dipolar energy is dominant again.

175

176 Finally, we show the thickness dependences of the maximum THE amplitudes
177 for non-hysteretic resistivity $\rho_{xy}^{\text{tTHE}}|_{\text{max}}$ and hysteretic resistivity $\rho_{xy}^{\text{hTHE}}|_{\text{max}}$, in Fig.
178 5(a) and Fig. 5(b), respectively. Notably, $\rho_{xy}^{\text{tTHE}}|_{\text{max}}$ increases from 100 nm to 40 nm
179 below $T < T_s$, from $0.1 \mu\Omega \text{ cm}$ to $0.6 \mu\Omega \text{ cm}$. Such a dependence of $\rho_{xy}^{\text{tTHE}}|_{\text{max}}$ is
180 comparable with that seen in previous reports on Mn_2PtSn thin films. Thick epitaxial
181 Mn_2PtSn film, i.e., close to the bulk condition, reported negligible THE, which may
182 reflect the inherent crystal structure. [35]. The largest THE more than $0.5 \mu\Omega \text{ cm}$ was
183 reported for polycrystalline Mn_2PtSn crystal growth on thermally oxidized silicon, with
184 the expectation of large defects and grains compared to epitaxial films. [16] Therefore, a

185 complete $I\bar{4}m2$ Mn_2PtSn thin film is considered to possess an inherently weak THE,
186 and a decrease in thickness breaks the c-axis crystal symmetry and enhances the eventual
187 THE. The thickness dependence of $\rho_{xy}^{\text{tTHE}}|_{\text{max}}$ above $T > T_s$ appears to be complicated,
188 showing weak but sign-inverted signals between 40 ~ 60 nm. We speculate incomplete
189 coplanar spins above $T > T_s$ may trigger such a non-systematic behavior for thinner film
190 samples. The detail study of the magnetic structure will be required to access the origin
191 of these anomalies.

192 The dependence of $\rho_{xy}^{\text{hTHE}}|_{\text{max}}$ is also clear in any temperature range. It shows
193 a monotonic increase from 100 nm to 40 nm for $T < T_s$ and decrease for $T > T_s$,
194 indicating the size and density of nontrivial spin textures, somewhen antiskyrmions,
195 shows direct and systematic dependences on both thickness and temperature. The
196 systematic variation of magnetic anisotropy of thin films is considered to play an essential
197 role here. These results are examples of thin-film engineering to modulate THE,
198 combined with stabilization of the antiskyrmion phase by tuning its dipolar energy.

199
200 In summary, we fabricated epitaxial tetragonal Heusler Mn_2PtSn thin films via
201 magnetron sputtering and observed a large THE. The condition of antiskyrmion
202 formations is systematically discussed in relation to the hysteretic properties of THE.
203 Antiskyrmion-type signal was observed at zero field vicinity for the thickest 100 nm thin
204 film at the noncoplanar ferrimagnetic phase. An additional antiskyrmion signal was
205 obtained for the thinnest 40nm film above the reorientation temperature, which is unique
206 to the thin film sample with weak magnetic anisotropy. These topological signals can be

207 systematically modulated by temperature and film thickness, providing a route to the
208 realization of skyrmion-based spintronic devices.

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ACKNOWLEDGEMENTS

211 This work was partially supported by the Japan Society for the Promotion of Science
212 (JSPS) KAKENHI grant nos. JP20K14419, JP17K18892, JP18H03787, and JST,
213 PRESTO grant no. JPMJPR18L3, Japan. The authors acknowledge W. Koshibae and X.
214 Z. Yu for fruitful discussions.

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DATA AVAILABILITY

217 The data that support the findings of this study are available within this article.

218

219

Figure Captions

220

221 FIG. 1. (a) Crystal structure of tetragonal Mn_2PtSn with $I\bar{4}m2$ space group. (b) X-ray
222 diffraction pattern of 40 nm Mn_2PtSn thin film. (c) Cross-sectional transmission electron
223 microscopy images of 100 nm and 40 nm Mn_2PtSn thin films. Positions of Mn, Pt, and
224 Sn atoms are modeled with red, green, and blue colors, respectively. (d) Nanobeam
225 electron diffraction pattern of 40 nm Mn_2PtSn thin film along [100] direction. (e)
226 Temperature dependences of longitudinal resistivities of Mn_2PtSn . The spin reorientation
227 temperatures T_s are indicated with dashed lines. (f) Temperature dependences of field-
228 cooled magnetization of Mn_2PtSn at perpendicular fields of 4 T.

229

230 FIG. 2. Hysteresis loops of ρ_{xy} (black), ρ_{xx} (green), $\rho_{xy}^{\text{OHE}} + \rho_{xy}^{\text{AHE}}$ (red), and ρ_{xy}^{THE}
231 (blue) of (a) 40 nm thick Mn_2PtSn at 150 K, (c) 40 nm thick Mn_2PtSn at 50 K, and (e)
232 100 nm thick Mn_2PtSn at 50 K, respectively. The closed (open) symbols indicate results
233 for the field-sweep from +4 T to -4 T (-4 T to +4 T). Field dependences of ρ_{xy}^{THE}
234 (black) and ρ_{xy}^{hTHE} (blue) of (b) 40 nm thick Mn_2PtSn at 150 K, (d) 40 nm thick Mn_2PtSn
235 at 50 K, and (f) 100 nm thick Mn_2PtSn at 50 K, respectively.

236

237 FIG. 3. Temperature and field dependences of ρ_{xy}^{THE} and ρ_{xy}^{hTHE} , for (a), (b) 100 nm
238 thick Mn_2PtSn film, (c), (d) 60 nm thick Mn_2PtSn film, and (e), (f) 40 nm thick
239 Mn_2PtSn film.

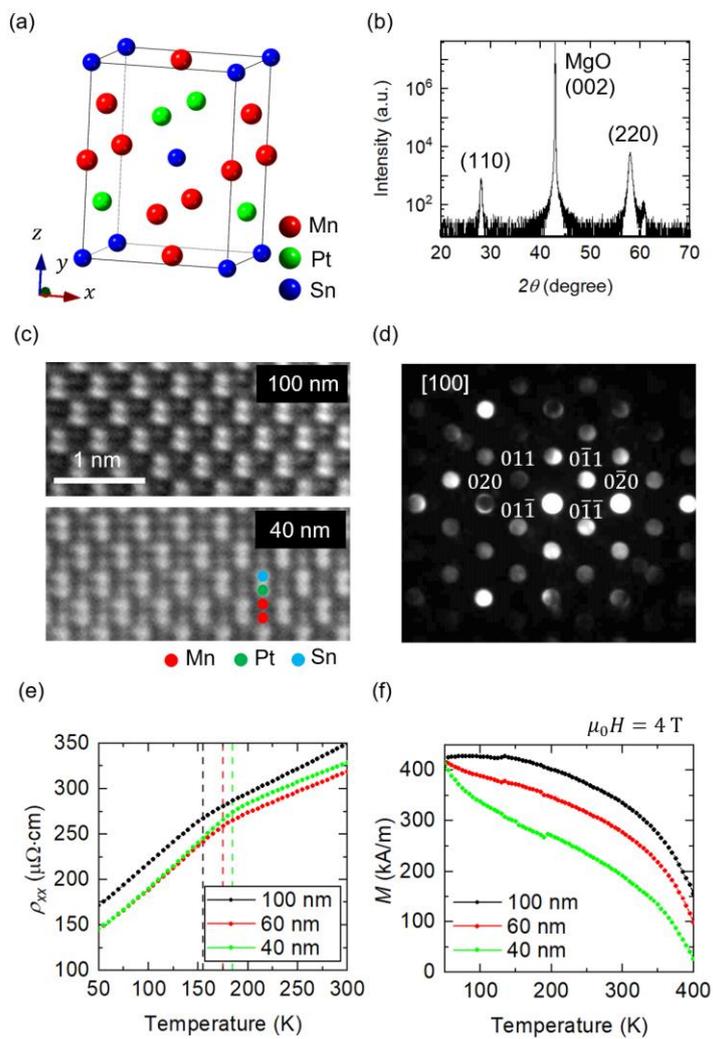
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241 FIG. 4. Film thickness dependences of $\rho_{xy}^{\text{tTHE}}|_{\text{max}}$ and $\rho_{xy}^{\text{hTHE}}|_{\text{max}}$ obtained (a) below
242 spin reorientation temperature $T < T_s$, and (b) above $T > T_s$. The black and red dotted
243 lines indicate the increase in $\rho_{xy}^{\text{tTHE}}|_{\text{max}}$ and $\rho_{xy}^{\text{hTHE}}|_{\text{max}}$.

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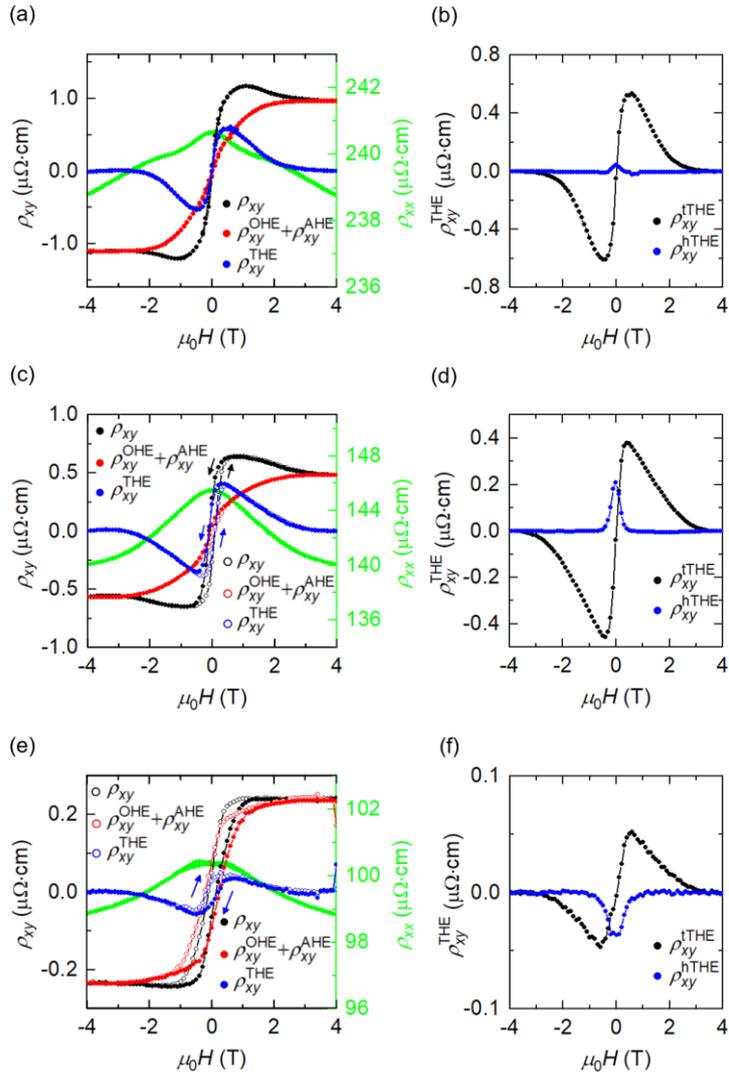
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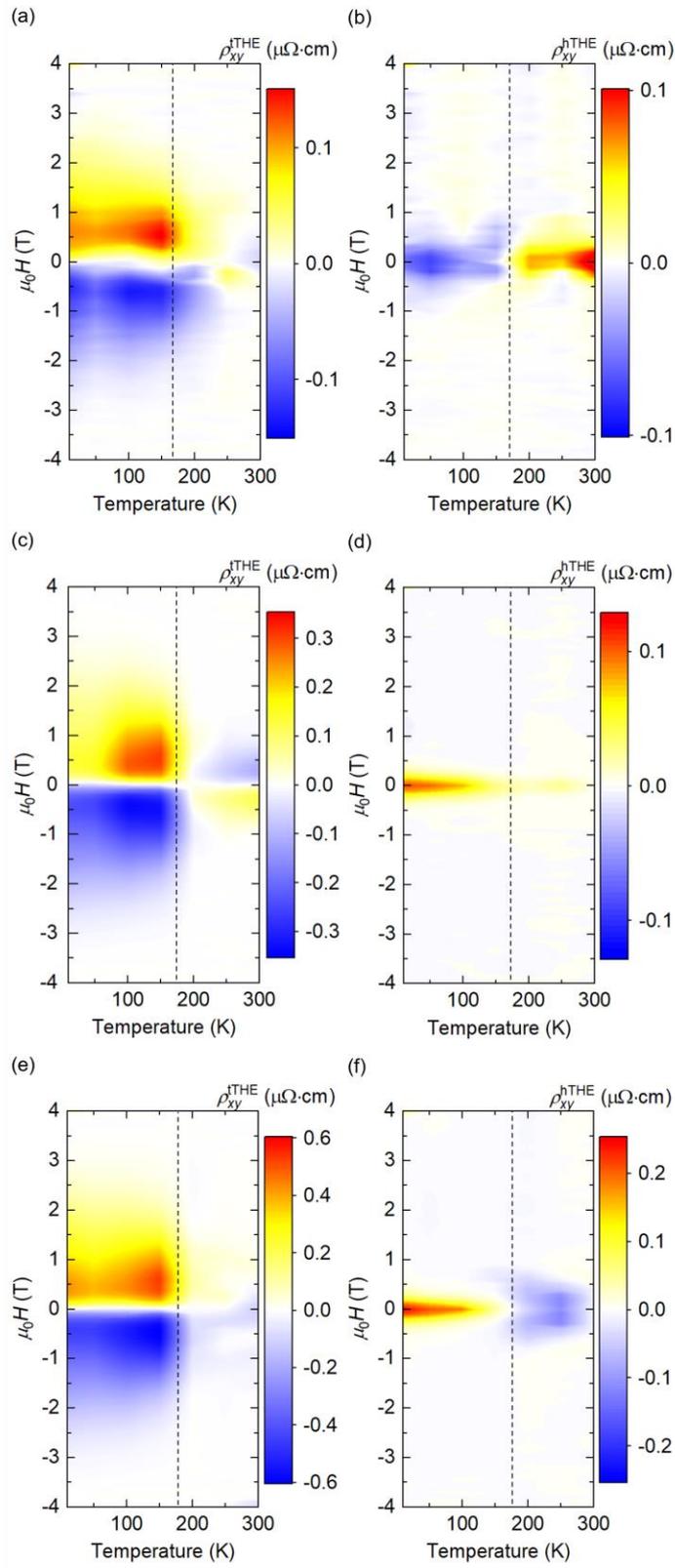
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Fig. 1 Sugimoto *et al.*



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Fig. 2 Sugimoto *et al.*



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Fig. 3 Sugimoto *et al.*

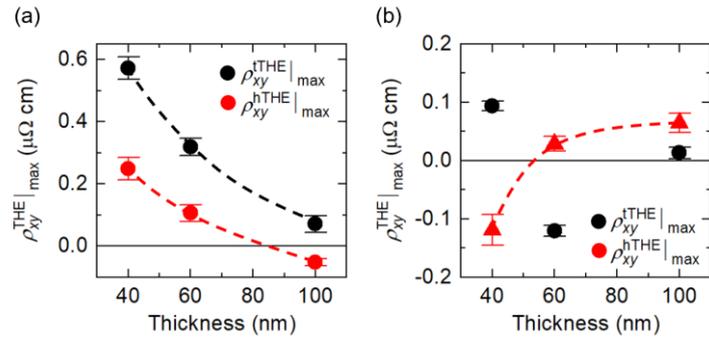


Fig. 4 Sugimoto *et al.*

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- 1 F. Jonietz, S. Mühlbauer, C. Pfleiderer, A. Neubauer, W. Münzer, A. Bauer, T. Adams, R. Georgii, P. Böni, R. A. Duine, K. Everschor, M. Garst, and A. Rosch, *Science* **330**, 1648 (2010).
- 2 A. Neubauer, C. Pfleiderer, B. Binz, A. Rosch, R. Ritz, P. G. Niklowitz, and P. Böni, *Phys. Rev. Lett.* **102**, 186602 (2009).
- 3 N. Kanazawa, Y. Onose, T. Arima, D. Okuyama, K. Ohoyama, S. Wakimoto, K. Kakurai, S. Ishiwata, and Y. Tokura, *Phys. Rev. Lett.* **106**, 156603 (2011).
- 4 M. Onoda, G. Tatara, and N. Nagaosa, *J. Phys. Soc. Jpn.* **73**, 2624 (2004).
- 5 A. Fert, V. Cros, and J. Sampaio, *Nat. Nanotechnol.* **8**, 152 (2013).
- 6 S. Mühlbauer, B. Binz, F. Jonietz, C. Pfleiderer, A. Rosch, A. Neubauer, R. Georgii, and P. Böni, *Science* **323**, 915 (2009).
- 7 X. Z. Yu, N. Kanazawa, Y. Onose, K. Kimoto, W. Z. Zhang, S. Ishiwata, Y. Matsui, and Y. Tokura, *Nat. Mater.* **10**, 106 (2011).
- 8 S. Ishiwata, M. Tokunaga, Y. Kaneko, D. Okuyama, Y. Tokunaga, S. Wakimoto, K. Kakurai, T. Arima, Y. Taguchi, and Y. Tokura, *Phys. Rev. B* **84**, 054427 (2011).
- 9 S. Chakraverty, T. Matsuda, H. Wadati, J. Okamoto, Y. Yamasaki, H. Nakao, Y. Murakami, S. Shiwata, M. Kawasaki, and Y. Taguchi, *Phys. Rev. B* **88**, 220405 (2013).
- 10 A. N. Bogdanov, U. K. Rößler, M. Wolf, and K.-H. Müller, *Phys. Rev. B* **66**, 214410 (2002).
- 11 W. Koshibae and N. Nagaosa, *Nat. Commun.* **7**, 10542 (2016).
- 12 A. K. Nayak, V. Kumar, T. Ma, P. Werner, E. Pippel, R. Sahoo, F. Damay, U. K. Rößler, C. Felser, and S. S. P. Parkin, *Nature (London)* **548**, 561 (2017).
- 13 S. Huang, C. Zhou, G. Chen, H. Shen, A. K. Schmid, K. Liu, and Y. Wu, *Phys. Rev. B* **96**, 144412 (2017).
- 14 Z. H. Liu, A. Burigu, Y. J. Zhang, H. M. Jafri, X. Q. Ma, E. K. Liu, W. H. Wang, and G. H. Wu, *Scr. Mater.* **143**, 122 (2018).
- 15 V. Kumar, N. Kumar, M. Reehuis, J. Gayles, A. S. Sukhanov, A. Hoser, F. Damay, C. Shekhar, P. Adler, and C. Felser, *Phys. Rev. B* **101**, 014424 (2020).
- 16 Y. Li, B. Ding, X. Wang, H. Zhang, W. Wang, and Z. Liu, *Appl. Phys. Lett.* **113**, 062406 (2018).

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- 17 P. Swekis, A. Markou, D. Kriegner, J. Gayles, R. Schlitz, W. Schnelle, S. T. B. Goennenwein, and C. Felser, *Phys. Rev. Mater.* **3**, 013001(R) (2019).
- 18 P. Swekis, J. Gayles, D. Kriegner, G. H. Fecher, Y. Sun, S. T. B. Goennenwein, C. Felser, and A. Markou, *ACS Appl. Electron. Mater.* **3**, 1323 (2021).
- 19 K. G. Rana, O. Meshcheriakova, J. Kübler, B. Ernst, J. Karel, R. Hillebrand, E. Pippel, P. Werner, A. K. Nayak, C. Felser, and S. S. P. Parkin, *New J. Phys.* **18**, 085007 (2016).
- 20 P. K. Sivakumar, B. Göbel, E. Lesne, A. Markou, J. Gidugu, J. M. Taylor, H. Deniz, J. Jena, C. Felser, I. Mertig, and S. S. P. Parkin, *ACS Nano* **14**, 13463 (2020).
- 21 Y. T. Y. Taguchi, Y. Oohara, H. Yoshizawa, N. Nagaosa, and Y. Tokura, *Science* **291**, 2573 (2001).
- 22 Y. Machida, S. Nakatsuji, Y. Maeno, T. Tayama, T. Sakakibara, and S. Onoda, *Phys. Rev. Lett.* **98**, 057203 (2007).
- 23 B. G. Ueland, C. F. Miclea, Y. Kato, A. Valenzuela, R. D. McDonald, R. Okazaki, P. H. Tobash, M. A. Torrez, F. Ronnig, R. Movshovich, Z. Fisk, E. D. Bauer, I. Martin, and J. D. Thompson, *Nat. Commun.* **3**, 1067 (2012).
- 24 L. Vistoli, W. Wang, A. Sander, Q. Zhu, B. Casals, R. Cichelero, A. Barthélémy, S. Fusil, G. Herranz, S. Valencia, R. Abrudan, E. Weschke, K. Nakazawa, H. Kohno, J. Santamaria, W. Wu, V. Garcia, and M. Bibes, *Nat. Phys.* **15**, 67 (2019).
- 25 J. J. Nakane, K. Nakazawa, and K. Nakazawa, *Phys. Rev. B* **101**, 174432 (2020).
- 26 N. Nagaosa, J. Sinova, S. Onoda, A. H. MacDonald, and N. P. Ong, *Rev. Mod. Phys.* **82**, 1539 (2010).
- 27 T. Miyasato, N. Abe, T. Fujii, A. Asamitsu, S. Onoda, Y. Onose, N. Nagaosa, and Y. Tokura, *Phys. Rev. Lett.* **99**, 086602 (2007).
- 28 V. Kumar, N. Kumar, M. Reehuis, J. Gayles, A. S. Sukhanov, A. Hoser, F. Damay, C. Shekhar, P. Adler, and C. Felser, *Phys. Rev. B* **101**, 014424 (2020).
- 29 N. Nagaosa, and Y. Tokura, *Nat. Nanotechnol.* **8**, 899 (2013).
- 30 A.A. Kovalev and S. Sandhoefner, *Front. Phys.* **6**, 98 (2018).
- 31 Y. F. Li, N. Kanazawa, X. Z. Yu, A. Tsukazaki, M. Kawasaki, M. Ichikawa, X. F. Jin, F. Kagawa, and Y. Tokura, *Phys. Rev. Lett.* **110**, 117202 (2013).
- 32 S. X. Huang and C. L. Chien, *Phys. Rev. Lett.* **108**, 267201 (2012).

33 K. Karube, L. Peng, J. Masell, X. Z. Yu, F. Kagawa, Y. Tokura, and Y. Taguchi, *Nat Mater.* **20**, 335 (2021).

34 M. Heigl, S. Koraltan, M. Vanatka, R. Kraft, and C. Abert, C. Vogler, A. Semisalova, P. Che, A. Ullrich, T. Schmidt, J. Hintermayr, D. Grundler, M. Farle, M. Urbánek, D. Suess, and M. Albrecht, *Nat. Commun.* **12**, 2611 (2021).

35 Y. Jin, S. Valloppilly, P. Kharel, J. Waybright, P. Lukashev, X. Z. Li, and D. J. Sellmyer, *J. Appl. Phys.* **124**, 103903 (2018).