



Article

Reliable Ohmic Contact Properties for Ni/Hydrogen-Terminated Diamond at Annealing Temperature up to 900 °C

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Abstract: Ohmic contact with high thermal stability is essential to promote hydrogen-terminated diamond (H-diamond) electronic devices for high-temperature applications. Here, the ohmic contact characteristics of Ni/H-diamond at annealing temperatures up to 900 °C are investigated. The measured current–voltage curves and deduced specific contact resistance (ρ_c) are used to evaluate the quality of the contact properties. Schottky contacts are formed for the as-received and 300 °C-annealed Ni/H-diamonds. When the annealing temperature is increased to 500 °C, the ohmic contact properties are formed with the ρ_c of $1.5 \times 10^{-3} \Omega \cdot \text{cm}^2$ for the Ni/H-diamond. As the annealing temperature rises to 900 °C, the ρ_c is determined to be as low as $6.0 \times 10^{-5} \Omega \cdot \text{cm}^2$. It is believed that the formation of Ni-related carbides at the Ni/H-diamond interface promotes the decrease in ρ_c . The Ni metal is extremely promising to be used as the ohmic contact electrode for the H-diamond-based electronic devices at temperature up to 900 °C.

Keywords: hydrogen-terminated diamond (H-diamond); ohmic contact; Ni; specific contact resistance; high-temperature



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1. Introduction

Diamond, with many remarkable intrinsic properties, possesses vast prospect applications for high-power, high-frequency, and high-temperature electronics [1–3]. It exhibits an ultrawide energy bandgap (5.5 eV), high carrier mobilities (4500 and 3800 $\text{cm}^2 \cdot \text{V}^{-1} \cdot \text{s}^{-1}$ for electrons and holes, respectively), large breakdown field strength (10 $\text{MV} \cdot \text{cm}^{-1}$), and the highest thermal conductivity (22 $\text{W} \cdot \text{cm}^{-1} \cdot \text{K}^{-1}$) [4]. Compared with boron-doped diamond, hydrogen-terminated diamond (H-diamond) shows outstanding p-type surface conductivity with a hole carrier concentration up to $\sim 10^{14} \text{ cm}^{-2}$ [5,6]. Recently, H-diamond-based field-effect transistors have achieved excellent device performances, such as a high breakdown voltage (2000 V), a high-output current density (1.3 $\text{A} \cdot \text{mm}^{-1}$), and a high-output power density (3.8 $\text{W} \cdot \text{mm}^{-1}$) [1,7,8]. Meanwhile, the passivation layer protection for the H-diamond surface improves the conductive stability of H-diamond-based electronic devices, even at temperatures as high as 500 °C [9–11]. The re-hydrogenation process enables the H-diamond surface damaged during annealing to regain good conductivity [12].

In order to further promote H-diamond-based electronic devices to be operated well at high temperatures, thermal-stable ohmic contact is essential. Until now, different kinds of metals are used for ohmic contacts on the H-diamond, such as, Au, Pd, Pt, W, Ti/Au, Pt/Au, Pd/Ti/Au, Ti/Ni/Au, etc. [12–15]. They show good ohmic contact properties and high thermal stability, with annealing temperatures up to 700 °C. On the other hand, the Ni metal tends to form Schottky contact with the H-diamond at an annealing temperature lower than 100 °C [16]. However, Ni, as a carbophilic element, is prone to react with carbon

at an elevated temperature [17]. The solid-solution reaction makes the Ni-related carbides formed at the Ni/H-diamond interface, which would possibly contribute to the formation of ohmic contact.

Here, ohmic contact characteristics of Ni/H-diamond at annealing temperatures up to 900 °C are investigated. The measured current–voltage curves and deduced specific contact resistance (ρ_C) are used to evaluate the quality of the contact properties.

2. Experimental

2.1. Preparation of H-Diamond Epitaxial Layer

An Ib-type (100) facet single-crystalline diamond was boiled in mixed H_2SO_4 and HNO_3 solutions at 300 °C for 3 h. Then, it was ultrasonically cleaned using acetone, ethanol, and deionized water sequentially. A 150-nm-thick H-diamond epitaxial layer was grown using a microwave plasma-enhanced chemical vapor deposition system. The CH_4 flow rate, H_2 flow rate, chamber pressure, and deposition temperature were 0.5 sccm, 500 sccm, 80 Torr, and 900–940 °C, respectively.

2.2. Formation of Transmission Line Model (TLM) Patterns for Ni on the H-Diamond

The H-diamond was sequentially coated with LOR5A and AZ5214E positive photoresists using a spin-coater with a rotation rate and time of 7000 rpm and 1 s, respectively. After exposing using a mask-less lithography system with a dose energy of $250 \text{ mJ}\cdot\text{cm}^{-2}$, the sample was developed in a 2.38% tetramethylammonium hydroxide solution for 90 s. The Ti metal used as key-patterns was deposited on the H-diamond by a J-sputter system in an Ar atmosphere in order to align the positions of the mesa structure and contact metals. The mesa structure was formed using a capacitively coupled plasma reactive-ion etching system. The plasma power, O_2 flow rate, and etching time were 50 W, 100 sccm, and 90 s, respectively. The five-group TLM electrode patterns were completed using the lithography procedures. The Ni, with a thickness of 100 nm, was grown on the H-diamond for ohmic contact via an e-beam evaporation system under a $\sim 10^{-5}$ Pa vacuum condition.

2.3. Annealing Process and Current–Voltage Measurements

The annealing process was performed using a rapid thermal annealing system in an Ar atmosphere. The annealing temperatures were 300, 500, 700, and 900 °C with an annealing time of 10 min for each temperature. After annealing, the sample was exposed to air for more than 10 h in order to promote the formation of a negatively adsorbed layer on the surface of the H-diamond, and to regain good surface conductivity. The calculations of ρ_C for the Ni/H-diamond with the TLM patterns can be referred to in the previous reports [18,19]. The electrical properties of Ni/H-diamond contacts were characterized by a room-temperature probe system.

3. Results and Discussion

Figure 1 shows the surface morphology of five-group TLM patterns of Ni on the H-diamond epitaxial layer before annealing. The length and width of each electrode are the same as 100 μm . All the Ni metals were stable to be formed on the H-diamond. The five groups of TLM patterns were used to characterize current–voltage curves for the Ni/H-diamonds of as-received, 300 °C-annealed, 500 °C-annealed, 700 °C-annealed, and 900 °C-annealed, respectively. The interspace (d) values from left to right in Figure 1, between the two adjacent electrodes for the five-group TLM patterns, are in the ranges of 7.9–9.1 μm , 13.5–13.9 μm , 18.0–18.8 μm , 23.1–23.9 μm and 28.2–29.1 μm , respectively.

Figure 2 shows current–voltage characteristics of (a) the as-received and (b) the 300 °C-annealed Ni/H-diamond contacts, respectively. The applied voltage is in the range of -1.0 – 1.0 V. For the as-received Ni/H-diamond contact, the output currents are in the order of 10^{-7} A (Figure 2a). All the curves show non-linear characteristics, indicating the Schottky contacts. After annealing at 300 °C for 10 min, as shown in Figure 2b, the output currents of the Ni/H-diamond increase to the order of 10^{-4} A. The annealing process improves

the contact properties of the Ni/H-diamond. However, all the curves still show non-linear characteristics. Therefore, even after annealing at 300 °C, the Ni/H-diamond still operates with Schottky contacts, which is consistent with the results obtained from the other report [20]. The presence of a chemisorbed species on the H-diamond surface possibly results in the formation of Schottky contact. However, the annealing process can improve the contact interface of the Ni/H-diamond, thereby promoting the current flow.

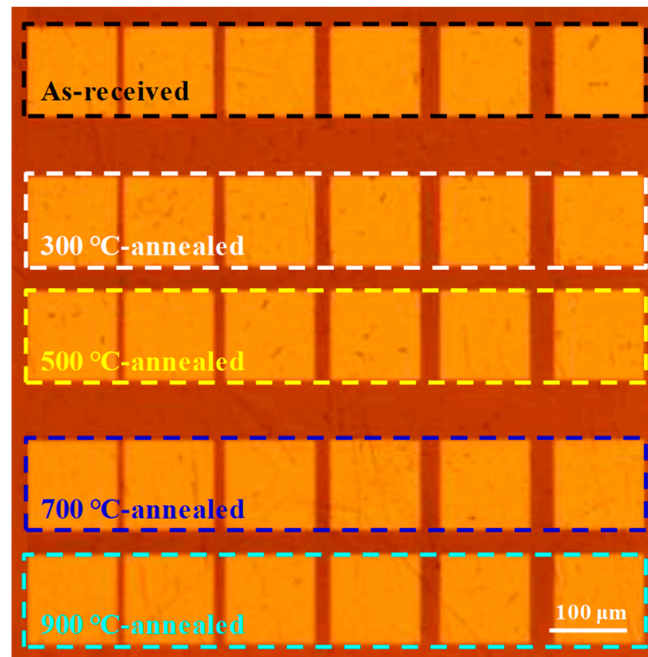


Figure 1. Surface morphology of the five-group transmission line model (TLM) patterns of Ni on the hydrogen-terminated diamond (H-diamond) epitaxial layer before annealing. They were used to characterize current–voltage curves for the Ni/H-diamonds of as-received, 300 °C-annealed, 500 °C-annealed, 700 °C-annealed, and 900 °C-annealed, respectively.

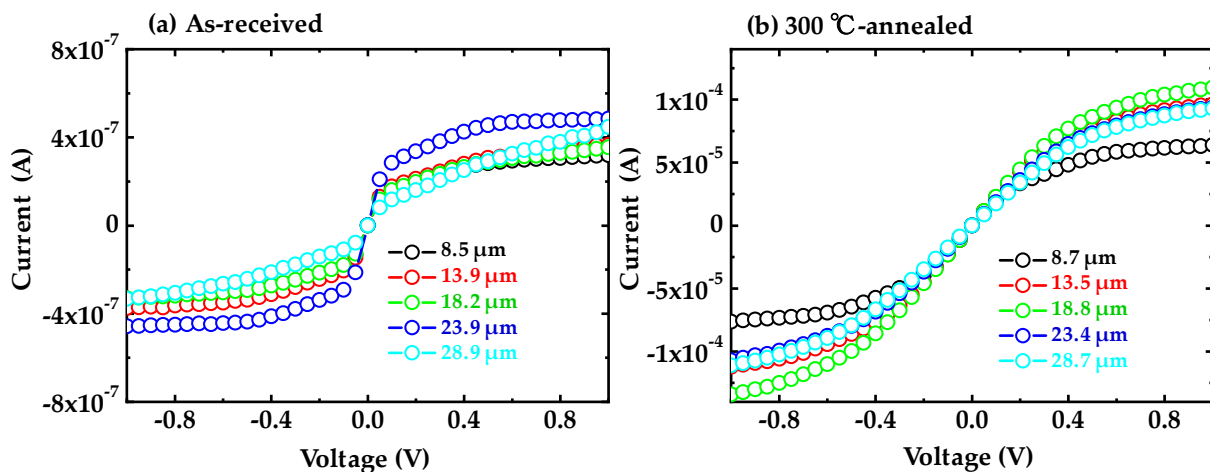


Figure 2. Current–voltage characteristics of (a) the as-received and (b) the 300 °C-annealed Ni/H-diamond contacts, respectively.

Figure 3a shows the current–voltage characteristics of the Ni/H-diamond after annealing at 500 °C. All the current–voltage curves have linear characteristics, which implies good ohmic contacts for the Ni/H-diamond. For two adjacent electrodes in which $d = 8.7 \mu\text{m}$, the output currents are $2.0 \times 10^{-4} \text{ A}$ at $\pm 1.0 \text{ V}$. The total resistance (R_T) for the 500 °C-

annealed Ni/H-diamond can be calculated to be $4.9 \times 10^3 \Omega$ ($d = 8.7 \mu\text{m}$). Based on the current–voltage characteristics for other adjacent electrodes (Figure 3a), the R_T for the Ni/H-diamond with other d values is also calculated and summarized in Figure 3b. There are the following relationships for the R_T and ρ_C with the Ni/H-diamond contact resistance (R_C), H-diamond surface sheet resistance (R_S), electrode transfer length (L_T), and electrode width (W) [18]:

$$R_T = 2R_C + \frac{R_S}{W}d \quad (1)$$

$$\rho_C = R_C \times L_T \times W \quad (2)$$

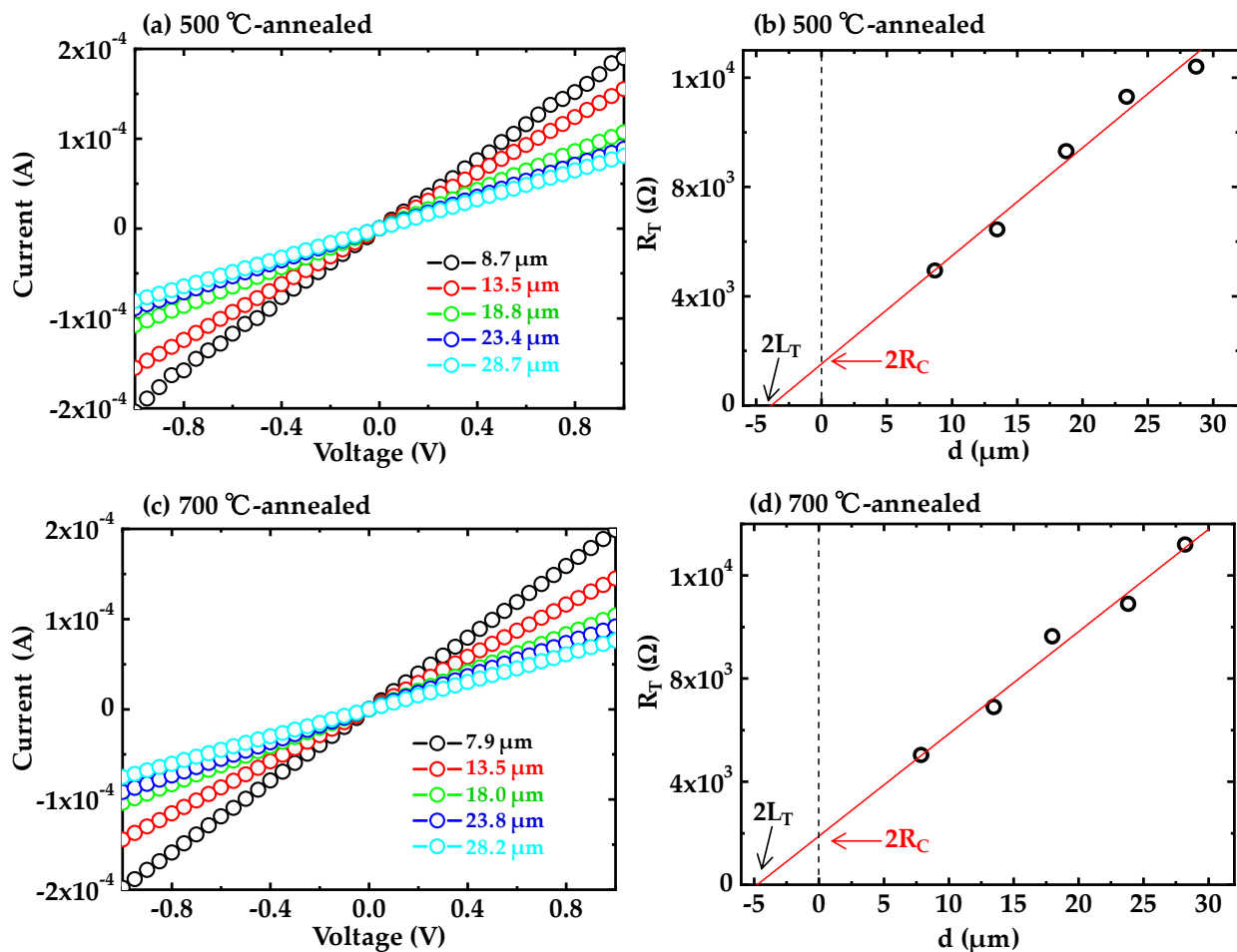


Figure 3. (a,c) Current–voltage characteristics of the 500 °C-annealed and 700 °C-annealed Ni/H-diamonds, respectively. (b,d) R_T as functions of d for the 500 °C-annealed and 700 °C-annealed Ni/H-diamonds, respectively.

By fitting the spots in Figure 3b, the R_S/W (the slope of fitting line) are determined to be $4.0 \times 10^2 \Omega \cdot \mu\text{m}^{-1}$ with an R_S of 40 k Ω . The $2R_C$ (the intercept of the y -axis) and $2L_T$ (the intercept of the x -axis) are deduced to be $1.5 \times 10^3 \Omega$ and 3.9 μm , respectively. Based on Equation (2), the ρ_C for the 500 °C-annealed Ni/H-diamond can be calculated to be $1.5 \times 10^{-3} \Omega \cdot \text{cm}^2$.

When the annealing temperature is increased to 700 °C, the output current maxima are the same level as those of the 500 °C-annealed Ni/H-diamond (Figure 3c). The linear characteristics for all the current–voltage curves are observed, indicating the good ohmic properties for the 700 °C-annealed Ni/H-diamond. Figure 3d shows the R_T as a function of d for the Ni/H-diamond after annealing at 700 °C. By fitting the spots, the R_S/W , $2R_C$, and $2L_T$ are obtained to be $4.0 \times 10^2 \Omega \cdot \mu\text{m}^{-1}$, $1.9 \times 10^3 \Omega$, and 4.8 μm , respectively. The R_S and ρ_C for the Ni/H-diamond after annealing at 700 °C are calculated to be

40 k Ω and $2.3 \times 10^{-3} \Omega \cdot \text{cm}^2$, respectively, which are close to those of the 500 °C-annealed Ni/H-diamond.

Figure 4a shows the current-voltage curves of the Ni/H-diamond after annealing at 900 °C. All the curves have good linear relationships. Therefore, the ohmic contacts of the Ni/H-diamond possess good thermal stability even after annealing at temperatures as high as 900 °C for 10 min, which is comparable with those of the Ni/SiC contacts [21,22].

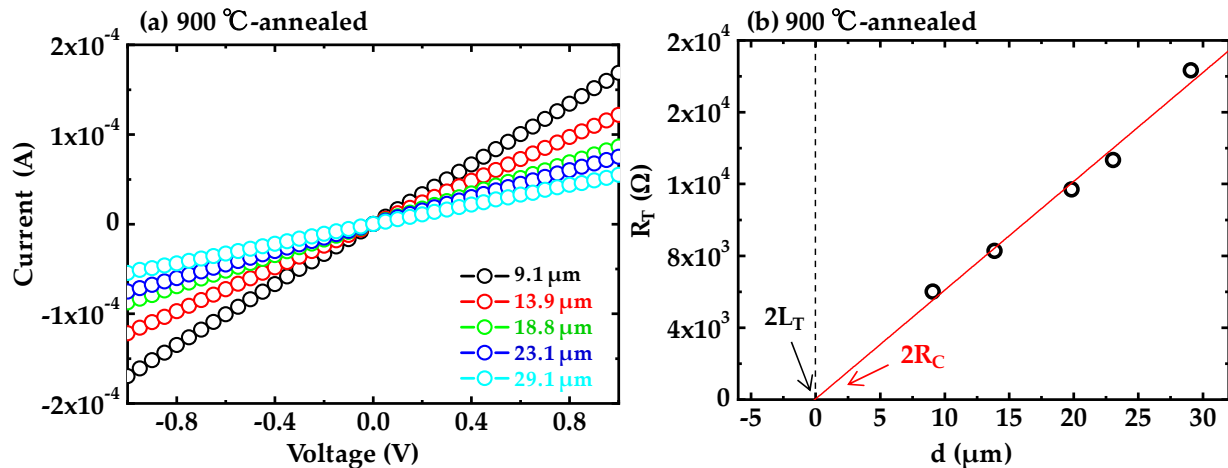


Figure 4. (a) Current–voltage characteristics and (b) total resistance (R_T) as functions of d for the 900 °C-annealed Ni/H-diamond.

Comparing with the output currents of the 500 °C-annealed and 700 °C-annealed Ni/H-diamonds at ± 1.0 V, those for the 900 °C-annealed one decreased slightly. Figure 4b shows the corresponding R_T as a function of d for the Ni/H-diamond after annealing at 900 °C. The R_S/W , $2R_C$, and L_T are deduced to be $6.1 \times 10^2 \Omega \cdot \mu\text{m}^{-1}$, 37.9 Ω , and 0.1 μm , respectively. The R_S and ρ_C are calculated to be 60.6 k Ω and $6.0 \times 10^{-5} \Omega \cdot \text{cm}^2$. The ρ_C is comparable with other metals on the H-diamond [23,24]. The thermal-stable Ni/H-diamond ohmic contacts exhibit advantages for the high-temperature application of H-diamond-based devices.

Figure 5a,b compare the R_C , R_S and ρ_C of the Ni/H-diamonds after annealing at 500, 700, and 900 °C, respectively. After annealing at 500 and 700 °C, the R_C and R_S values of the Ni/H-diamonds show no obvious changes. After annealing at 900 °C, however, the R_C decreases to 19.0 Ω , while the R_S increases to 60.6 k Ω . The increased R_S is possibly attributed to the damage of C–H bonds or the desorbed absorption layer on the H-diamond surface after multiplied high-temperature treatments [5]. The decrease in R_C may be due to the carbon phase transition at the interface between Ni and H-diamond at high temperatures [24,25]. Under the catalysis of nickel, diamond is prone to transform into the graphite phase or form carbide related with Ni, which greatly increases the electrical conductivity, thereby greatly reducing the contact resistance at the interface.

In order to confirm the interface reaction between Ni and H-diamond after annealing, transmission electron microscope (TEM) (Figure 6a) and energy dispersive spectrometer (EDS) (Figure 6b) measurements for the Ni/H-diamond after annealing were performed. The interface for the Ni/H-diamond is ambiguous and curved after annealing. The EDS result shows a great number of carbon atoms from diamond dissolved into the Ni lattice. Therefore, the carbides related with Ni for the Ni/H-diamond are formed at the interface, which leads to a lower ρ_C .

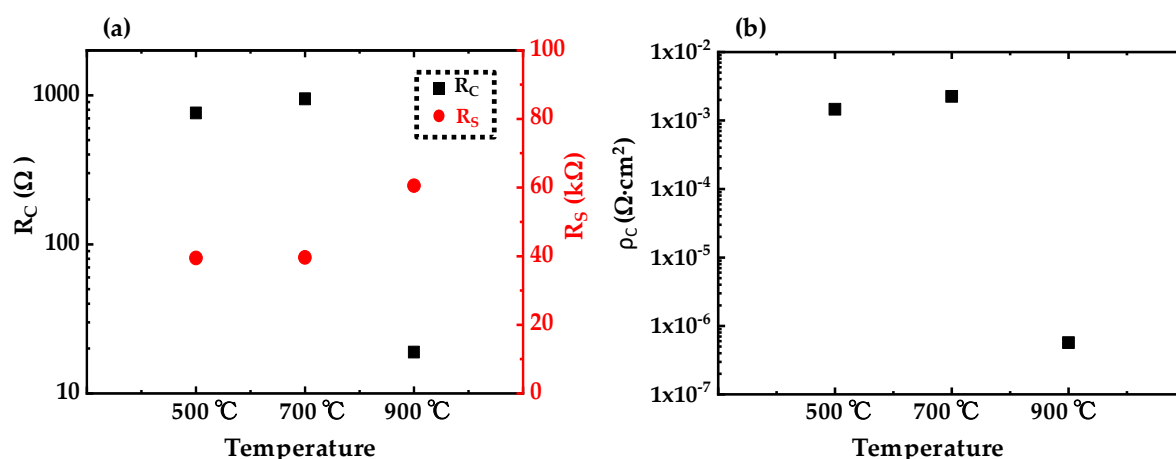


Figure 5. Summary of (a) the contact resistance (R_C) and surface sheet resistance (R_S), and (b) the deduced specific contact resistance (ρ_C) for the Ni/H-diamond after annealing at 500, 700, and 900 °C, respectively.

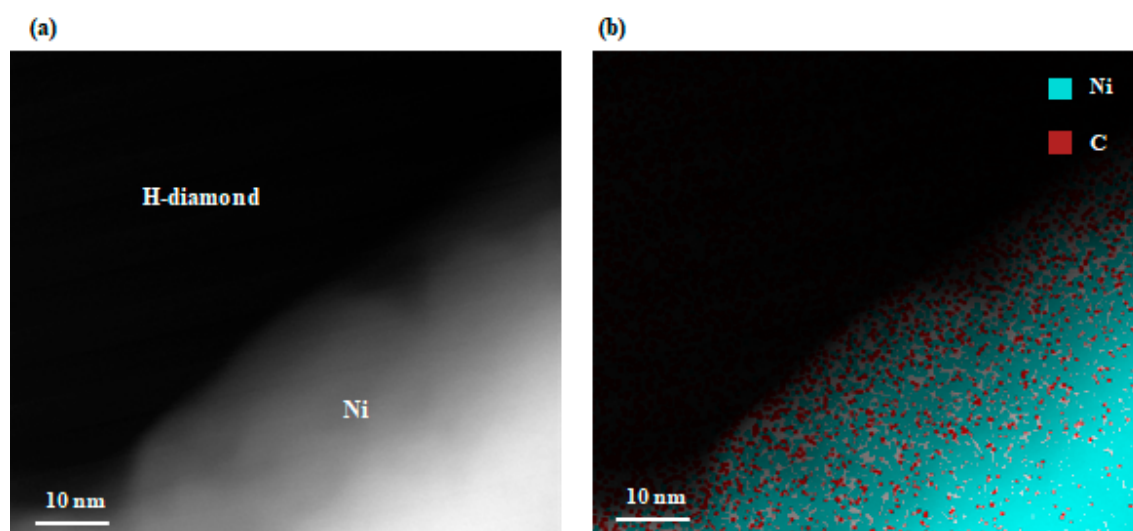


Figure 6. (a) Transmission electron microscope (TEM) and (b) energy dispersive spectrometer (EDS) images for the interface of the Ni/H-diamond after annealing at 900 °C, respectively.

4. Conclusions

In this study, ohmic contact characteristics of the Ni/H-diamond at annealing temperatures up to 900 °C were investigated. Schottky contacts were formed for the as-received and the 300 °C-annealed Ni/H-diamonds. When the annealing temperatures were increased to 500 °C, ohmic contacts were formed with the ρ_C of $1.5 \times 10^{-3} \Omega \cdot \text{cm}^2$ for the Ni/H-diamond. For the 700 °C-annealed Ni/H-diamond, the ρ_C was the same level as that of the 500 °C-annealed one. As the annealing temperature rose to 900 °C, the specific contact resistance was as low as $6.0 \times 10^{-5} \Omega \cdot \text{cm}^2$. It is believed that the formation of Ni-related carbides at the Ni/H-diamond interface promoted the decrease in specific contact resistance. Therefore, the thermal-stable Ni/H-diamond ohmic contacts exhibited advantages for the high-temperature application of H-diamond-based devices.

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